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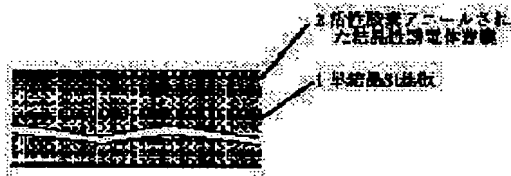
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(54) COMPOSITE STRUCTURE OF CRYSTALLINE OXIDE DIELECTRIC THIN FILM AND SINGLE CRYSTAL SILICON SUBSTRATE, ELECTRONIC DEVICE USING THE SAME, AND MANUFACTURE OF THEREOF

(57)Abstract:

PROBLEM TO BE SOLVED: To realize an IS structure having a crystalline oxide dielectric layer having little leak current, a high breakdown voltage, and a sufficiently long life with the elapse of time against the breakdown voltage, that is, a composite structure of a crystalline oxide dielectric thin film and a single crystal silicon substrate.

SOLUTION: In a composite structure in which a crystalline oxide dielectric thin film 2 is grown on a single crystal silicon semiconductor substrate 1, the crystalline oxide dielectric thin film 2 is formed by a thin film (single film or stacked layer made of, for example, stabilized zirconia, cerium oxide, strontium titanate, magnesium oxide, or yttrium oxide) to which a heat treatment is performed in an active oxygen atmosphere after the growth.



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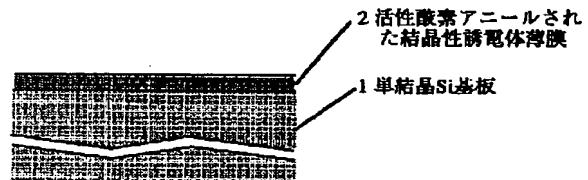
(54) 【発明の名称】 結晶性酸化物誘電体薄膜と単結晶シリコン基体との複合構造体およびそれを用いた電子素子およびそれらの製造方法

(57) 【要約】

【課題】 リーク電流が少なく、絶縁破壊耐圧が高く、経時的絶縁破壊寿命が充分長い結晶性酸化物誘電体層を有する I S 構造、すなわち結晶性酸化物誘電体薄膜と単結晶シリコン基体との複合構造体を実現する。

【解決手段】 単結晶シリコン半導体基体 1 に結晶性酸化物誘電体薄膜 2 を成長させた複合構造体において、前記結晶性酸化物誘電体薄膜 2 を、成長後に活性酸素雰囲気中で熱処理を施された薄膜（例えば安定化ジルコニア、酸化セリウム、チタン酸ストロンチウム、酸化マグネシウム、酸化イットリウムの何れかからなる単層膜または積層膜）で形成した複合構造体。

(図 1)



【特許請求の範囲】

【請求項 1】単結晶シリコン半導体基体に結晶性酸化物誘電体薄膜を成長させた複合構造体において、前記結晶性酸化物誘電体薄膜が、成長後に活性酸素雰囲気中で熱処理を施された薄膜であることを特徴とする結晶性酸化物誘電体薄膜と単結晶シリコン基体との複合構造体。

【請求項 2】単結晶シリコン半導体基体に結晶性酸化物誘電体薄膜を成長させた複合構造体を含む電子素子において、

前記複合構造体が、その形成後から電子素子構造を完成させるまでの間に少なくとも 1 回、活性酸素雰囲気中で熱処理を施された薄膜であることを特徴とする電子素子。

【請求項 3】前記結晶性酸化物誘電体薄膜が前記単結晶シリコン基体にヘテロエピタキシャル成長させた単結晶膜であることを特徴とする請求項 1 に記載の複合構造体または請求項 2 に記載の電子素子。

【請求項 4】前記結晶性酸化物誘電体薄膜が複数の材料の積層によって形成された多層膜からなることを特徴とする請求項 1 乃至請求項 3 の何れかに記載の複合構造体または電子素子。

【請求項 5】前記結晶性酸化物誘電体薄膜が安定化ジルコニア、酸化セリウム、チタン酸ストロンチウム、酸化マグネシウム、酸化イットリウムから選ばれた一つあるいは複数の材料によって構成された単層膜あるいは積層膜であることを特徴とする請求項 1 乃至請求項 4 の何れかに記載の複合構造体または電子素子。

【請求項 6】上記電子素子は、単結晶シリコン半導体基体 (S) の一部に、順に結晶性酸化物誘電体薄膜 (I) と強誘電体薄膜 (F) と金属電極膜 (M) を積層し、該積層膜の非形成部の少なくとも一部にソース電極とドレイン電極とを設けた MFIS トランジスタであり、該結晶性酸化物薄膜が、成長後もしくは強誘電体薄膜成長後に、少なくとも 1 回、活性酸素雰囲気中で熱処理を施された薄膜であることを特徴とする請求項 2 乃至請求項 5 の何れかに記載の電子素子。

【請求項 7】上記電子素子は、単結晶シリコン半導体基体 (S) の上に一つまたは複数の結晶性酸化物誘電体薄膜 (I) と単結晶シリコン膜をエピタキシャル成長させた SOI 基板であり、該結晶性酸化物薄膜が、その成長後に、少なくとも 1 回、活性酸素雰囲気中で熱処理を施された薄膜であることを特徴とする請求項 2 乃至請求項 5 の何れかに記載の電子素子。

【請求項 8】前記活性酸素雰囲気での熱処理が、酸素を供給しながら 190 nm 未満の波長を含む紫外線照射下で熱処理する方法、オゾンを経由しながら熱処理する方法、酸素プラズマ中で熱処理する方法、のいずれか一つの方法、あるいは二つ以上の方法を組合せて同時に行なう方法、あるいは二つ以上の方法を異なった時点で重複

して行なう方法によって達成されることを特徴とする請求項 1 乃至請求項 7 の何れかに記載の複合構造体または電子素子の製造方法。

【請求項 9】前記活性酸素雰囲気での熱処理が結晶性酸化物誘電体薄膜形成直後に実施されることを特徴とする請求項 1 乃至請求項 8 の何れかに記載の複合構造体または電子素子の製造方法。

【請求項 10】前記活性酸素雰囲気での熱処理が結晶性酸化物誘電体薄膜を形成し、該結晶性酸化物誘電体薄膜の上に導電性酸化物、あるいは、酸化物半導体膜、酸化物超電導体膜、酸化物強誘電体膜の少なくとも一つを形成した後に施されることを特徴とする請求項 2 乃至請求項 8 の何れかに記載の電子素子の製造方法。

【発明の詳細な説明】

【0001】

【発明の属する技術分野】本発明は、リーク電流の極めて少ない結晶性の酸化物誘電体薄膜と単結晶シリコン (Si) 基板との複合構造体とこれを利用した電子素子の構造、ならびにこれら構造体や電子素子の製造方法に関する。なお、ここで言う結晶性誘電体薄膜とは単結晶または特定の方向に優先配向した誘電体薄膜を意味するものとする。

【0002】

【従来の技術】単結晶膜は、一般に、アモルファス膜や多結晶膜に比べて、結晶構造的に平坦かつ均一なため、導電性、誘電性、絶縁性、強誘電性、超電導性、半導体特性、光透過性、光磁気特性、結晶周期性等のその材料特有の物理的諸特性が強く鋭く現れるという特長を有している。そのため、様々な工業分野で有用であり、その実現が強く望まれている。近年、これら単結晶膜を Si 半導体素子や集積回路に積載して、従来の Si 系材料だけでは達成できない付加価値の高い集積回路やスマートデバイスを実現しようとする試みが活発に行われるようになってきている。このような新規な Si 半導体デバイスを構成する上で欠かせない基本構造が結晶性の誘電体薄膜 (I) と単結晶 Si (S) との複合構造体 (以下 IS 構造体と略称) である。

【0003】一例を挙げると、図 8 の模式的構造断面図に示すような MFIS 型不揮発性ランダムアクセスメモリセルがある。このセルは金属 (M) - シリコン酸化膜 (O) - 半導体 (S) からなる MOS トランジスタの O 部分を、電気的に反転可能な自発分極をもつ単結晶強誘電体膜 (F) と単結晶誘電体膜 (I) とで置き換えたものである。

【0004】図 8 において、101 は p 型の (100) 面の単結晶 Si 基板、102 は Si 基板 101 上にヘテロエピタキシャル成長された単結晶酸化物誘電体膜 (I) であり、たとえば (100) 面の YSZ 膜である。103 は単結晶酸化物誘電体膜 102 の上にヘテロエピタキシャル成長された強誘電体膜であり、たとえば

(001)面の $PbTiO_3$ 膜である。104は強誘電体膜103上に形成されたゲート電極であり、たとえばPt電極である。105および106はn型のソース電極およびドレイン電極であり、それぞれP(リン)やAs(ヒ素)をイオン注入して、熱拡散で活性化させて形成したものである。

【0005】このメモリでは、ゲート電極(G)-Si基板(Sub)間に正または負のパルス電圧を印加して、強誘電体膜の自発分極ベクトルを反転固定させ、トランジスタを導通あるいは非導通状態にさせることで情報の記録を行う。

【0006】また、上記の強誘電体層としては、化学的气相成長法(CVD)やマグネトロンスパッタリング法などで形成された $PbTiO_3$ (チタン酸鉛)膜や $Pb(Zr_xTi_{1-x})O_3$ (ジルコンチタン酸鉛)膜、 $Bi_4Ti_3O_{12}$ (チタン酸ビスマス)膜など(いずれも分極軸配向単結晶膜)が検討されている。

【0007】また、上記の単結晶誘電体層としては、電子ビーム蒸着法やCVDで形成された CeO_2 (酸化セリウム)膜やYSZ(イットリヤ安定化ジルコニヤ)膜、 $SrTiO_3$ (チタン酸ストロンチウム)膜などが検討されている。これらの単結晶誘電体膜は単結晶強誘電体膜を形成するためのテンプレート層として機能するほかに、強誘電体とSi基板が相互拡散するのを防止する緩衝層としての役割を担っている。

【0008】また、不揮発性ランダムアクセスメモリセルには、上記のMFIS構造の他に、MFMIS構造も提案されている。ここで単結晶F層と単結晶I層の間に挿入されたM層は単結晶の導電膜、たとえばPt膜などである。この場合もMFMIS構造の下部にIS構造体が用いられる。

【0009】次に、図9の模式的構造断面図に示したSOI構造のMOSトランジスタもIS構造体の重要な応用例である。この構造のMOSトランジスタは動作上極めて有害なラッチアップ現象から完全に免れることができるという特筆すべき特徴がある。

【0010】図9において、111は(100)面のSi基板、112は該Si基板111に接してヘテロエピタキシャル成長された CeO_2 やYSZ、 $SrTiO_3$ などの単結晶誘電体膜である。このSi基板111と単結晶誘電体膜112とがIS構造を成している。113は単結晶誘電体膜112上にCVDでヘテロエピタキシャル成長されたp型の(100)面の単結晶Si層、114は単結晶Si層113の熱酸化で形成された SiO_2 ゲート酸化膜、115はゲート酸化膜114の上にCVDとドライエッチングで形成されたポリシリコンのゲート電極である。116および117はN形のソース電極およびドレイン電極であり、それぞれP(リン)やAs(ヒ素)をイオン注入して、熱拡散で活性化させて形成したものである。

【0011】また、集積回路の高速化処理を目指して試みられている酸化物高温超電導薄膜配線などでもIS構造体の上に高品質の単結晶超電導体膜(たとえば $YBa_2Cu_3O_7$)配線を形成する。I層としては、たとえばYSZと Y_2O_3 の単結晶積層膜が用いられる。

【0012】このようにIS構造体は、単結晶の機能性薄膜(強誘電体膜、半導体膜、導電膜、超電導膜、誘電体膜)を含有する様々なデバイスを単結晶Si基板上で実現するための重要な基本構造であることが理解される。しかし、現状では単結晶酸化物誘電体薄膜I層に、

(1)リーク電流が大きい、(2)破壊強度が低い、(3)経時破壊(TDDB)寿命が短い、という問題があり、上述のような高機能デバイスの実現に大きな障害となっている。

【0013】上記の問題を実際のデータに基づいて詳しく説明する。図7は結晶性酸化物誘電体薄膜のリーク電流特性図である。図7において、特性曲線(a)は従来の単結晶YSZ膜の特性例を示すものであり、N型単結晶(100)面のSi基板にエピタキシャル成長させた(100)面のYSZ膜に、直径200[μm]のAl電極(ゲート)を形成して作製したMIS容量のゲート電極に、正の電圧を印加して測定したリーク電流密度 J [A/cm²]を、電界強度 E_{ox} [V/cm]の関数として示したものである。なお、吸収電流成分を除くために、電流の測定は、電圧印加してから充分時間が経過した後に行っている。また、膜厚は40[nm]であり、成膜法は、ここでは電子ビーム蒸着法である。基板の洗浄法ならびに成膜法(蒸着条件等)は本発明の実施例の中で後述する。

【0014】図7の特性曲線(a)を見てわかるように、従来例の単結晶YSZ膜のリーク電流は非常に高い水準にある。実用を考えると、リーク電流密度は電界強度 $E_{ox}=1$ [MV/cm]のとき、 $J=1$ [nA/cm²]以下に抑制したい。しかし、上記の従来例では $J=100$ [μA/cm²]台と、これと大きくかけ離れた値を示している。また、図7の特性曲線(a)におけるA点の電流ジャンプは、この電圧($E_{ox} \approx 1.5$ [MV/cm])でYSZ膜が絶縁破壊をしていることを示している。前述のMFIS不揮発性メモリセルなどに適用するには、この絶縁破壊強度は充分とは言えない値である。

【0015】さらにI層には、電源ストレスに対する長期的な絶縁破壊(TDDB)耐性があることが必須の条件である。この耐性は一般に誘電体膜に定電流ストレスを与え、絶縁破壊するまでに膜を通過した総電荷量 Q_{bd} [C/cm²]を測定して評価する。上記の単結晶YSZ膜に、 $J=0.1$ [mA/cm²]の定電流ストレスを与えた時の典型的なTDDB寿命は、それぞれ $Q_{bd}=1.2$ [mC/cm²]であった。この値はMOSトランジスタなどのゲート酸化膜として用いられている熱酸

化SiO₂膜の代表的な値Q_{bd}~10 [C/cm²]と並べると比べようもなく低い。電源ストレスが比較的強いMFI S不揮発性メモリセルなどにYSZ膜を適用するには、少なくとも0.1 [C/cm²] 台のTDDB寿命が必要である。

【0016】IS構造におけるこのようなI層の問題は、電子ビーム蒸着法で形成した単結晶YSZ膜に限ったことではなく、他の単結晶酸化物誘電体材料膜や他の成膜法で作製したYSZ膜でも共通して観察される問題である。

【0017】

【発明が解決しようとする課題】本発明は、このような従来のIS構造体およびこれを用いたデバイスの結晶性誘電体膜（I層）の問題点を解決するためになされたものであり、リーク電流が少なく、絶縁破壊耐圧が高く、経時的絶縁破壊寿命が充分長い結晶性酸化物誘電体層を有するIS構造、すなわち結晶性酸化物誘電体薄膜と単結晶シリコン基板との複合構造体およびこれを用いた電子素子およびそれらの製造方法を実現することを目的としている。

【0018】

【課題を解決するための手段】上記目的を達成するために、本発明においては、IS複合構造体およびこれを用いた電子素子およびそれらの製造方法を各特許請求の範囲に記載しているように構成した。すなわち、請求項1に記載の発明においては、単結晶シリコン半導体基体に結晶性酸化物誘電体薄膜を成長させたIS複合構造体において、I層は成長後に活性酸素雰囲気中で熱処理を施した結晶性酸化物誘電体薄膜であるとした。

【0019】同様に請求項2においては、IS複合構造体を含み、これの上に構築された電子素子において、I層は成長後から電子素子を完成させるまでの間に活性酸素雰囲気中で熱処理（以下活性酸素アニールと称する）を施された結晶性酸化物誘電体薄膜であるとした。

【0020】請求項3~5は請求項1または請求項2の具体的な態様を示したものである。すなわち請求項3においては、結晶性酸化物誘電体薄膜を単結晶シリコン基体にヘテロエピタキシャル成長させた単結晶膜で構成している。また、請求項4においては前記結晶性酸化物誘電体薄膜を複数の酸化物誘電体膜の積層によって形成される多層膜によって構成している。また、請求項5においては結晶性酸化物誘電体薄膜が安定化ジルコニア、酸化セリウム、チタン酸ストロンチウム、酸化マグネシウム、酸化イットリウムから選ばれた単層膜あるいは積層複合膜であるとしている。

【0021】また、請求項6および請求項7は、電子素子の具体的な態様を示すものであり、請求項6は電子素子がMFI Sトランジスタ（MFI S型不揮発性ランダムアクセスメモリセル）の場合、請求項7は電子素子がS O I（Silicon On Insulator）基板である場合を示す。

【0022】さらに、請求項8は前記請求項に記載の複合構造体あるいは電子素子の製造方法であって、製造工程中にあって結晶性酸化物誘電体薄膜の活性酸素アニールは、酸素を供給しながら190nm未満の波長を含む紫外線照射下で熱処理する方法、オゾンを提供しながら熱処理する方法、酸素プラズマ下で熱処理する方法、のいずれか一つの方法、あるいは二つ以上の方法を組合せて同時に行なう方法（例えば紫外線照射下でオゾンを供給しながら熱処理する）、あるいは二つ以上の方法を異なった時点で重複して行なう方法（一つの方法を行なった後に次の方法を行なう）によって達成される。なお、上記紫外線の波長190nm未満とは、酸素を分解して活性酸素とする能力を有する範囲を意味する。

【0023】また、請求項9は、製造方法の一態様であって、活性酸素雰囲気での熱処理が結晶性酸化物誘電体薄膜形成直後に実施されるものである。また、請求項10は、前記電子素子の製造方法の一態様であって、活性酸素雰囲気での熱処理が結晶性酸化物誘電体薄膜を形成し、該結晶性酸化物誘電体薄膜の上に導電性酸化物、あるいは、酸化物半導体膜、酸化物超電導体膜、酸化物強誘電体膜の少なくとも一つを形成した後に施されるものである。

【0024】以下、作用について説明する。まず、単結晶Si基板上に形成した結晶性酸化物薄膜のリーク電流の機構について説明する。本発明者が鋭意研究して明らかにしたところによると、結晶性酸化物薄膜のリーク電流の原因には、（1）酸化物の結晶粒内部の酸素空位、（2）結晶粒内部の金属元素ボイド、（3）結晶粒界面のエネルギー準位、（4）結晶粒間隙の金属Si元素、などがあり、各原因に基づいて（分離はできないが）異なるリーク電流が流れる。上記の原因（1）と（2）は結晶の格子点にあるべき原子が確率的に抜けてしまういわゆる点欠陥である。金属元素が抜けるとこれと結合している酸素原子も一緒に抜けることが多く、金属-酸素ボイドとなる。これら点欠陥は酸化物の禁止帯に電子準位を形成し、この密度が一定以上になると準位間を容易に電子が移動できるようになり、所謂ホッピング電導を引き起こす。このような点欠陥はアモルファス膜でもしばしば観察され、やはりリーク電流の原因になっている。

【0025】上記の原因（1）、（2）が粒界内を流れるリーク電流の原因であったのに対して、原因（3）と（4）は粒界面を流れるリーク電流に関するものであり、結晶性酸化膜特有の原因である。ここで単結晶Siにエピタキシャル成長している結晶性酸化物薄膜に“粒界”とは、奇異な印象を受けかもしれないが、これは格子定数が微妙に違う異種材料を単結晶Siにヘテロエピタキシャル成長させるとき、ヘテロ面の格子不整ストレスを開放するために必然的に生じる粒界であり、成長面に垂直方向に粒界面が延びている。この粒界面では結晶

周期が中断し、粒界面に存在する原子のなかには、化学結合していないものが多数存在する。上記の原因(3)はこれら未結合原子が存在する結果として結晶性酸化物禁制帯内部に形成された電子準位である。電子準位が高濃度になるとバンドを形成しリーク電流を増大させる。原因(4)は結晶性酸化物薄膜の成膜時にSi基板から侵入し結晶粒界に偏析した金属Si原子である。結晶粒界には1～数原子層程度の空隙があるので、減圧下(低酸素濃度下)成長時に基板から侵入してきたSiは金属状態(または低価数状態)のまま粒界間隙にトラップされる。粒界に偏析したSiは当然、リーク電流の導電経路になる。

【0026】リーク電流の発生のメカニズムを説明し終えたところで、本発明がリーク電流を含む従来の問題を如何にして解決するか(作用)について説明する。本発明者の最近の研究成果によると、本発明によるリーク電流の低減はつぎのような機構に基づいて起こると考えられる。本発明の活性酸素アニールでは、反応性の高いO(¹D)やO₂(¹Δ_g)、O(³P)などの活性酸素が気相で豊富に発生している雰囲気中で結晶性酸化物薄膜/単結晶Si構造体を比較的低温で熱処理する。このとき活性酸素はSi基板に向かって内方拡散する。活性酸素の一部は原因(1)の酸素空位に遭遇し、空位を消滅(充填)させて安定する。また活性酸素の他の一部は粒界に向かって、あるいは、粒界に沿って拡散し、粒界に偏析している金属状シリコンを酸化してSiO₂とし、粒界間隙部をSiO₂ネットワークで満たす。これによって原因(4)の金属Siが除去される。なお、上記の熱処理温度は、常温から約500℃程度までの範囲とし、特に200～500℃程度が望ましい。その理由は500℃以上の高温では活性酸素が通常の酸素に変わってしまい、また、200℃以下ではアニール時間が長くなるためである。

【0027】さらに、結晶粒界面に存在していた未結合手原子Mは粒界間隙にSiO₂ネットワークが形成される過程で、近接するSiとM-O-Si結合を成し、その未結合手を解消する。これによって原因(3)が解消される。残りの活性酸素は単結晶Si基板まで拡散し、結晶性酸化物薄膜/Si界面に極めて薄いSiO₂膜を成長させる。SiO₂の成長は一般に体積の膨張を伴うが、熱処理温度が低温である活性酸素アニールでは、通常の熱酸化のように圧縮応力に打ち勝って成長するのは困難であるから、活性酸素アニールでは一部のSi元素を酸化物膜中に放出し、体積の膨張を押えながら界面での酸化が進行する。ここで放出されたSi原子は外方拡散する過程で金属-酸素ボイドに遭遇し、金属原子位置に捕獲される。一方、このとき気相からは活性酸素が内方拡散しており、ボイドに捕獲されたSi原子に遭遇することが可能であり、ボイドに近接する酸化物の複数の金属とSi原子との中間に入ってM-O-Si結合を完

成させる。こうして原因(2)の金属-酸素ボイドはSi-O_x(x=4～5)原子団によって補填され、消滅する。なお、ボイドに捕獲されなかった大多数のSi原子は酸化物薄膜の表面に析出する。この析出物は結晶性酸化物薄膜上への他の薄膜のエピタキシャル成長を阻害するので、本発明においては成長前に弗酸溶液などで除去される。

【0028】こうして本発明の構造ならびに方法によれば、リーク電流の上記原因(1)～(4)をすべて除去することができる。この結果、本発明は実施例にて後述するように、従来例に比べ、結晶性酸化物薄膜のリーク電流を数桁のレベルで低減することができる。

【0029】つぎに本発明による絶縁破壊強度BE_{ox}やTDD寿命を示す総電荷量Q_{bd}の改善について述べる。本発明の対象とするCeO₂やYSZなどの結晶性酸化物薄膜の絶縁破壊や疲労のメカニズムは、研究段階にあり十分に解明されているとはいえない。しかし、電界強度E_{ox}や総電荷量Q_{bd}とリーク電流Jとの間には強い正の相関があることは広く認められている。これは、E_{ox}やQ_{bd}を決定する要因とリーク電流の原因とが共通していることを示唆している。本発明は上述のように結晶性酸化物のリーク電流の原因を除去することができるから、これと深く関わる絶縁破壊や疲労の要因を取り除くことができ、結果として、従来例に比べて、絶縁破壊強度BE_{ox}を高く、総電荷量Q_{bd}を大きくすることができる。

【0030】

【発明の効果】以上説明してきたように本発明においては、結晶性酸化物誘電体薄膜と単結晶Si基板との複合構造体において、該結晶性誘電体薄膜を成長後に活性酸素アニールを施した膜としたため、結晶性誘電体膜における(1)リーク電流特性、(2)絶縁破壊強度、

(3)経時的絶縁破壊(TDD)耐性、を改善することができるという効果が得られる。

【0031】また本発明においては、結晶性酸化物誘電体膜と単結晶Si基板との複合構造体を含む電子素子上記本発明による複合構造体を適用することにより、上記(1)～(3)の特性不良に起因して生じていた電子素子の破壊や動作不良を解消できるという効果が得られる。

【0032】

【発明の実施の形態】以下、本発明にかかる結晶性酸化物誘電体薄膜と単結晶Si基板との複合構造体と、該複合構造体を含む電子素子と、複合構造体および電子素子の製造方法を、実施の形態および実施例に基づいて説明する。

【0033】(実施の形態)図1は本発明に基づく結晶性酸化物誘電体薄膜と単結晶Si基板との複合構造体の一実施の形態を示す断面図である。図1において、1は(100)、(111)、(110)面などの所定の結

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晶面を有する単結晶 Si 基板、2 は、電子ビーム蒸着法、イオンビームスパッタリング法、分子線エピタキシー法、レーザアブレーション法などの物理的気相成長法や CVD などの化学的気相成長法にて Si 基板 1 上にヘテロエピタキシャル成長された結晶性酸化物誘電体薄膜、あるいは単結晶酸化物誘電体薄膜である。これに該当する誘電体薄膜としては、 Y_2O_3 や Sm_2O_3 、 CaO など安定化されたジルコニア膜（たとえば YSZ 膜）、酸化セリウム (CeO_2) 膜、チタン酸ストロンチウム ($SrTiO_3$) 膜、酸化イットリウム (Y_2O_3) 膜、酸化マグネシウム (MgO) などが挙げられるが、これに限定されるものではなく、Si 基板にヘテロエピタキシャル成長される酸化物誘電体薄膜なら他の薄膜でもよい。酸化物誘電体薄膜 2 は、Si 基板 1 に成長された後、Si 基板 1 とともに O_3 や O (1P)、 O (1D)、 O_2 ($^1\Delta_g$)、 O_2^- などの活性酸素雰囲気の中で常温から 500 [°C] 未満の温度で所定時間、熱処理を施される。こうして本発明に基づく複合構造体が完成する。

【0034】以下に、上記の本発明複合構造体の実施の形態の具体的な実施例を提示するとともに、これを作製する工程を示すことによって本発明の製造方法の実施例も併せて説明することにする。

【0035】（実施例 1）本発明に基づく複合構造体の最初の具体的な実施例は (100) 面の単結晶 Si 基板 1 とエピタキシャル成長された (100) 面の単結晶イットリヤ安定化ジルコニア (YSZ) 薄膜 2 とからなる複合構造体を、高真空電子ビーム蒸着法と UV/ O_2 活性酸素アニールとを用いて構築する例である。ここで UV は活性酸素の生成手段としての紫外線を意味し、 O_2 は活性酸素の原料としての酸素を意味する。

【0036】はじめに、(100) 面の単結晶 Si 基板 1 の表面の汚染物および自然酸化物を、RCA 洗浄 (ア *

* シンモニア水と過酸化水素水の混合液による洗浄と、塩酸と過酸化水素水の混合液による洗浄からなる伝統的な Si 基板洗浄法) と希フッ酸洗浄 (5% 濃度の HF 水溶液に数 10 秒浸漬した後、純水でリンスし乾燥する洗浄法) とで除去する。

【0037】つづいて Si 基板 1 上に約 13 モル% の Y_2O_3 を含有する YSZ 膜を高真空電子ビーム蒸着法にて形成する。蒸着機は実験用あるいは量産用として市販されている汎用の高真空型の装置である。蒸着のターゲットは単結晶 YSZ (Y_2O_3 13 mol%) インゴットを直径 15 mm、厚み 7 mm の切削加工した YSZ タブレットである。

【0038】成膜の手順を説明すると、蒸着源に YSZ タブレットを充填し、洗浄した Si 基板 1 を蒸着槽の基板ホルダに設置して、速やかに蒸着槽内を排気する。基板ホルダは基板を常温から 1000 [°C] までの温度に加熱する能力を有しており、蒸着源とは 30 [cm] の距離を隔てて対抗している。蒸着装置の排気は強力であり、内部の圧力は 10^{-9} [Torr] 以下まで減圧することが可能である。蒸着槽の圧力が 10^{-6} [Torr] に達したら基板温度を加熱して所定の成膜温度にする。温度が安定し、かつ、蒸着槽内の圧力が 10^{-9} [Torr] 以下に下がったところで高速の電子ビームをタブレットに射突させ、0.2 [nm/min] の成長速度で YSZ 膜を Si 基板に成長させる。YSZ の膜厚が概ね 3 [nm] になったところで外部から蒸着槽に高純度乾燥酸素を導入して圧力を 2×10^{-6} [Torr] に調節するとともに成長速度を 5 [nm/min] に上げ、所定の膜厚になるまで成長を続ける。成長速度は熱陰極の放出電流を制御して調節する。(100) 面の単結晶 YSZ 薄膜が得られる典型的な成膜条件は次のとおりである。

【0039】

YSZ 成膜条件	
成膜圧力	1×10^{-8} [Torr] (膜厚 3 nm まで) 2×10^{-6} [Torr] (膜厚 3 nm から O_2 導入)
成膜温度	800 [°C] (基板温度)
ターゲット	$Y_2O_3 : ZrO_2 = 13 : 87$
基板距離	30 cm
電子ビーム加速電圧	8 kV

膜厚が所望の値になったところで蒸着電源の出力を切って成膜を終える。この後、蒸着槽に基板を置いたまま基板を徐冷し、基板温度が十分低くなってから基板を蒸着槽から取出す。このようにして堆積した YSZ 膜はホタル石構造 (立方晶) を有する単結晶エピタキシャル膜であった。

【0040】従来例として前に説明した YSZ 薄膜と単結晶 Si との複合構造体は、ここまでの工程で完成とな

る。しかし、本発明に基づく複合構造体ではこの後、活性酸素アニール工程を実施する。ここでは活性酸素アニールのひとつ、UV/ O_2 アニールの例で説明するが、後述の他の活性酸素アニールでも構わない。

【0041】図 2 は、アニールに使用する装置の模式的要部断面図である。図 2 において、11 は側壁が水冷されている反応器、12 は反応器 11 を排気するための真空ポンプで、反応器 11 の圧力を常圧から 10^{-6} [Torr]

rr]までの間に維持することができる。13は単結晶Si基板に単結晶YSZ薄膜を成長したアニール前の複合構造体である。14は13を支持し所定の温度に保持するためのサセプタである。複合構造体13はサセプタ14上にYSZ膜面が上に向くように置かれている。15は純度99.999%以上の酸素O₂を充填したガスボンベである。アニールに際して、ボンベのO₂は整圧器16で0.5[kg/cm²]程度の圧力に減圧された後、質量流量調節器17で所定の流量に調節され、反応器11に導かれる。18はキセノンガスが封入された誘電体バリア放電エキシマランプで、中心波長172[nm]の強力な紫外光UVを発生することができる。エキシマランプ18から出射された紫外励起線は合成石英窓19を通して反応器11内部に入射される。20は反応器11と真空ポンプ12を結ぶ排気配管を開閉する排気バルブ、21は大気排出バルブである。なお、上記紫外光の波長は酸素を分解して活性酸素とする能力を有する範囲の値であり、一般に190nm未満であることが必要である。

【0042】次のような手順でUV/O₂アニールを行 * 20

*う。所定の温度に加熱されているサセプタ14の上に複合構造体13を載せ、真空ポンプ12を作動させると同時に、排気バルブ20を開口し、直ちに反応器11内を真空排気する。内部の圧力が10⁻⁵[Torr]台に到達したところで排気バルブ20を閉口し、真空ポンプ12を停止し、高純度O₂ガスを反応器11に1000[cc/min]程度の質量流量で導入する。反応器の内圧が常圧(=大気圧)になったところで、大気排出バルブ21を開口し、以降導入される余剰なO₂ガスを所定の質量流量で反応器11外に排出する。こうすることによって反応器11を大気圧の純酸素雰囲気に維持することができる。つづいてエキシマランプ18から反応器11内部に紫外励起線を出射する。反応器11内部に入射した172[nm]波長の紫外光は反応器11内のO₂に強く吸収され、結果として、O₃やO(³P)、O(¹D)、O₂(¹Δg)、などからなる活性酸素が発生する。以上の準備が整ったところで、UV/O₂活性酸素アニールを所定の時間、実施する。典型的なアニール条件を下に示す。

【0043】

活性酸素アニール(UV/O₂)条件

容器圧力	大気圧
処理温度	350[°C]
励起強度(合成石英ガラス位置)	13[mW/cm ²](172[nm])
O ₂ 流量	200[cc/min]
処理時間	15[min]

アニールが終了したら、エキシマランプ18の出射と高純度O₂の導入を停止し、直ちに複合構造体13を取り出す。こうして、UV/O₂アニールが終わり、本発明の結晶性酸化物誘電体薄膜と単結晶Si基板との複合構造体が完成する。

【0044】なお、上記の処理温度(アニール温度)は、常温から約500℃程度までの範囲とし、特に200~500℃程度が望ましい。実験結果によると500℃以上の高温では活性酸素が通常酸素に変わってしまい、また、200℃以下ではアニール時間が長くなる。

【0045】次に、本発明の実施例の効果を従来例の実測データと対比させながら説明する。図7の特性曲線(b)は、上記実施例1に基づいて作製したYSZ/Si基板の複合構造体において、YSZ膜の上にAlゲート電極を蒸着して形成したMIS容量に正の電圧を印加して測定したリーク電流(YSZ膜に流れる電流)特性を示している。なお、特性曲線(a)は従来例による複合構造体のリーク特性(既述)である。

【0046】図7の特性曲線(a)と(b)を比較すれば、本発明の実施例が従来例に比べてリーク電流を概ね4桁~5桁低減させていることが分かる。E_{ox}=1[MV/cm]の特性を見ると、従来例が電流密度J=

100[μA/cm²]台なのに対し、本発明の実施例ではJ=1[nA/cm²]以下の値を与えている。

【0047】次に絶縁破壊強度BE_{ox}に着目すると、測定によれば、従来例ではBE_{ox}=1.5[MV/cm]であったが、本発明実施例ではBE_{ox}=4.6[MV/cm]と3倍以上の向上が見られた。

【0048】また、定電流ストレス0.1[mA/cm²]を与えた時のTDDB寿命は、従来例が総電荷量Q_{bd}=1.2[mC/cm²]であった(既述)のに対して、本発明の実施例ではQ_{bd}=850[mC/cm²]の値であって、3桁近い改善が得られた。

【0049】以上の結果から、本発明が、結晶性酸化物薄膜/Si基板の複合構造体における結晶性酸化物薄膜の、(1)リーク電流の低減、(2)絶縁破壊強度の向上、(3)TDDB寿命の増長、に極めて有効であることが理解される。

【0050】(実施例2)本発明に基づく複合構造体の2番目の具体的な実施例は、(100)面の単結晶Si基板とエピタキシャル成長させた(110)面の単結晶酸化セリウム(CeO₂)薄膜2とからなる複合構造体を、高真空電子ビーム蒸着法とUV/O₃活性酸素アニールを用いて形成する例である。ここで「/O₃」は

8%のオゾン O_3 を含む原料としての酸素を意味する。

【0051】はじめに、n型の(100)面の単結晶Si基板の表面の汚染物および自然酸化物をRCA洗浄と希フッ酸洗浄とで除去する。つづいてSi基板上に CeO_2 薄膜を高真空電子ビーム蒸着法にてピタキシャル成長させる。蒸着機は前記YSZ成長で用いた装置と同じものを用いる。蒸着ターゲットは純度99.999%以上の CeO_2 粉末をホットプレスで直径15[mm]、厚み7[mm]の形に成形したしたタブレットである。蒸発源に CeO_2 タブレットを充填し、Si基板を蒸着槽の基板ホルダに設置し、蒸着槽内を排気する。蒸着槽の圧力が 10^{-6} [Torr]に達したら基板温度を加熱*

*して所定の成膜温度にする。温度が安定し、槽内圧力が 10^{-9} [Torr]以下に下がったところで高速の電子ビームをタブレットに射突させ0.4[nm/min]の成長速度で CeO_2 膜を成長させる。膜厚が概ね5[nm]になったところで蒸着槽に高純度乾燥酸素を導入して圧力を 8×10^{-6} [Torr]に調節するとともに成長速度を1.55[nm/min]に上げ、所定の膜厚になるまで成長を続ける。成長速度は熱陰極の放出電流を制御して調節する。上記(110)面の単結晶 CeO_2 薄膜が得られる典型的な成膜条件は次のとおりである。

【0052】

(110)面の CeO_2 成膜条件

成膜圧力	1×10^{-8} [Torr] 以下 (膜厚3 [nm] まで)
	8×10^{-6} [Torr] (膜厚3 [nm] から O_2 導入)
成長温度	800 [°C] (基板温度)
ターゲット	純度99.999% CeO_2 タブレット
基板距離	30 cm
電子ビーム加速電圧	8 kV

所望の膜厚になったところで蒸着電源の出力を切って成膜を終える。このようにして堆積した CeO_2 膜はホタル石構造(立方晶)を有する(110)面の単結晶エピタキシャル膜であった。ちなみに、同じ(100)面のSi基板上に配向方向が異なる(111)面の単結晶 CeO_2 膜を得ることも可能である。この場合には、上記成長条件において、成長温度を600[°C]とする。従来例はこの段階で完成となる。本発明に基づく複合構造体ではこの後、活性酸素アニール「UV/ O_3 アニール」工程を実施する。

【0053】図3はアニールに使用する装置の模式的要部断面図である。図2と同じ番号を付した部位の機能・構成は図2と同じであるので、冗長を避け、説明を省略する。図3において、31は工程途上の CeO_2 /Si複合構造体である。整圧器16と質量流量計17の間には無声放電型のオゾン発生器32が設置されている。このオゾン発生器32は酸素を原料に8モル%の O_3 を発生することができる。33は低圧水銀ランプで、波長185[nm]と254[nm]に強い輝線をもつ紫外光を発生することができる。低圧水銀ランプ33から出射された紫外励起線は合成石英窓19を通して反応器11

内部に入射される。

【0054】UV/ O_3 アニールの手順は前述のUV/ O_2 アニールとほぼ同じである。すなわち所定温度のサセプタ14の上に構造体31を載せ、直ちに反応器11内を真空排気する。反応器11内部の圧力が 10^{-5} [Torr]台に到達したところで排気バルブ20を閉口し、真空ポンプ12を停止し、高純度 O_2 ガスを反応器11に1000[cc/min]程度の質量流量で導入する。反応器11の内圧が常圧(=大気圧)になったところで、大気排出バルブ21を開口すると同時にオゾン発生器32を作動させて8%の O_3 を含む高純度 O_2 ガスを反応器11に導入する。間を置かず低圧水銀ランプ33から反応器11内部に紫外励起線を出射する。反応器11内部に入射した254[nm]の紫外光は器内の O_3 を光分解し、結果として、 O_3 のほかに $O(^1D)$ 、 $O_2(^1\Delta_g)$ 、 $O(^3P)$ などからなる活性酸素が発生する。こうして、UV/ O_3 活性酸素アニールを所定の時間、実施する。典型的なアニール条件は次のとおりである。

【0055】

活性酸素アニール (UV/ O_3) 条件

容器圧力	大気圧
処理温度	400 [°C]
励起強度 (合成石英ガラス位置)	20 [mW/cm ²] (254 [nm])
O_2 流量	200 [cc/min]
処理時間	60 [min]

アニールが終了したら、低圧水銀ランプ33の出射とオゾン発生器32の作動、高純度 O_2 の導入を停止し、直ちに複合構造体を取り出す。こうして、本発明による結晶性酸化物誘電体薄膜と単結晶Si基板との複合構造体が完成する。

【0056】次に、本実施例2の効果を説明する。下記の表1は本実施例2ならびに従来例に基づく CeO_2 /*
(表1)

CeO_2 /Si複合構造体の電気特性の比較

	リーク電流JL [nA/cm ²]	絶縁破壊強度BEox [MV/cm]	総電荷量Qbd [mC/cm ²]
実施例	0.45	5.1	1800
従来例	3500	1.4	12.4

【0058】上記表1の電気特性は前記実施例1と同様にMIS容量(A1ゲート)に正の電圧を印加して測定した。表1から明かなとおり、本発明の実施例2は従来例に比べてリーク電流JLを概ね4桁低減し、絶縁破壊強度BEoxを3.5倍向上させている。さらにTDD B寿命に対応する総電荷量Qbdを1C/cm²台に乗せ、TDD B寿命を従来よりも2桁向上させることに成功した。以上の結果から、本発明の実施例2が、結晶性酸化物薄膜/Si基板の複合構造体における結晶性酸化物薄膜のリーク電流や絶縁破壊強度、TDD B寿命の改善に効果的であることは明白である。

【0059】なお、上記活性酸素アニールにおいて、UVを用いないアニール、すなわち O_3 アニールに置き換えることも可能である。 O_3 アニールも活性酸素を生じるので有効である。熱処理装置の構成はUV/ O_3 より簡単になるが、熱処理時間がUVを用いたときに比べて2倍程度のびる。また、前述実施例や後述の実施例の他の活性酸素アニール法も用いることができる。

【0060】(実施例3)本発明に基づく複合構造体の3番目の具体的な実施例は(100)面の単結晶Si基板1と(001)面の単結晶チタン酸ストロンチウム($SrTiO_3$)薄膜2とからなる複合構造体をレーザアブレーション蒸着法とplasma/ O_2 アニールで形成する例である。

【0061】n型の(100)面の単結晶Si基板の表面の汚染物および自然酸化物をRCA洗浄と希フッ酸洗浄とで除去した後、Si基板上に $SrTiO_3$ 薄膜を高真空レーザアブレーション蒸着法にて成長する。蒸着機は真空中で切り替え可能な2個の蒸着源を備えている汎用機である。各々の蒸着源には純度99.999%以上

*Si複合構造体の CeO_2 薄膜(20[nm])の電界強度1[MV/cm]におけるリーク電流JL、絶縁破壊強度BEox、定電流0.1[mA/cm²]を与えたときのTDD B寿命(総電荷量Qbd)を比較した表である。

【0057】

【表1】

の粉末をホットプレスで円盤状(10φ、10t)に整形した SrO タブレットと $SrTiO_3$ タブレットが充填されている。

【0062】成長の手順を説明すると、前記洗浄直後のSi基板1を蒸着槽中の基板ホルダに設置し、蒸着槽内を排気する。このホルダは基板を常温から800[℃]までの温度に加熱する能力を有しており、タブレットから約5[cm]の距離を挟んで対抗している。蒸着装置の排気は強力であり、内部の圧力は 10^{-9} [Torr]以下まで背圧を下げるのが可能である。蒸着槽の圧力が 10^{-8} [Torr]に達したところで基板ホルダを加熱して所定の成長温度にする。温度が安定するまでの間に、 SrO が充填されている第1の蒸発源を使用する準備を行う。

【0063】蒸着槽内の圧力が 10^{-8} [Torr]以下に下がったところでレーザ光源の出力をオンして高密度のレーザ光をタブレットに射突させ、非常に薄い SrO バッファ膜の成長を開始する。本実施例のレーザはKrF(波長248nm、パルス幅20ns)エキシマレーザである。繰り返し周波数は10[Hz]、エネルギー密度は1.0[J/cm²]であった。 SrO バッファ膜の厚みが6[nm]になったところでレーザの出力を停止して、成長を止める。直ちに蒸発源を $SrTiO_3$ タブレットが収められている第2の蒸発源に切り替えるとともに、基板ホルダの温度を $SrTiO_3$ 成長に適した所定の温度に変更する。温度が安定したところで、外部から高純度酸素を導入し、圧力を 5×10^{-5} [Torr]とし、再びレーザの出力をオンして今度は $SrTiO_3$ 膜を成長させる。膜厚が所望の値になったところでレーザ光源の発信を停止し、 O_2 の導入を止め、基板を

徐冷する。単結晶(001)面の SrTiO_3 膜が得られる成長条件の一例をまとめると次のとおりである。 *

SrO バッファ膜の成長条件

成長圧力 1×10^{-8} [Torr] 以下
 成長温度 800 [°C] (基板温度)
 タブレット-基板距離 5 [cm]
 レーザ光源 KrFエキシマレーザ (エネルギー130mJ)
 励起パルス 幅20ns、繰り返し周波数10Hz

SrTiO_3 膜の成長条件

成長圧力 5×10^{-5} [Torr] (99.999%酸素導入)
 成長温度 600 [°C] (基板温度)
 他の条件は上の SrO と同じ

基板温度が十分低くなってから基板を蒸着槽から取出す。最初に形成した6[nm]の SrO 層は後の SrTiO_3 膜を成長する過程で消失する(SrTiO_3 膜に変質したと考えられる)ので、こうして形成した膜は事実上単結晶 SrTiO_3 単層膜である。

【0065】従来例はこの段階の構成で完成である。本発明に基づく複合構造体ではこの後、活性酸素アニールを行う。前出の UV/O_2 アニールや UV/O_3 アニールでも可能であるが、ここでは「plasma/ O_2 アニール」の例を紹介する。

【0066】図4はplasma/ O_2 アニールに使用する装置の模式的要部断面図である。図4において、41は側壁が水冷されている反応器、42は反応器41を排気するための真空ポンプで、反応器41の圧力を常圧から 10^{-6} [Torr]までの間に維持することができる。43はアニール前の複合構造体である。44は複合構造体43を支持し所定の温度に保持する加熱手段を備えたサセプタである。このサセプタ44は電気的に接地されている。複合構造体43は SrTiO_3 膜面が上に向くようにサセプタ44に置かれる。サセプタ44と対抗するように置かれているのがパワー電極45である。パワー電極45はブロッキング容量46と整合器47を介して器外の高周波電源48(例えば周波数13.56MHz)に電気的に接続されている。49は純度99. ※

※99.9%以上の酸素 O_2 を充填したガスボンベである。

ガスボンベ49の O_2 は整圧器50で0.5[kg/cm²]程度の圧力に減圧された後、質量流量調節器51で所定の流量に調節され、反応器41に導かれる。52は反応器41と真空ポンプ42を結ぶ排気配管を開閉する排気バルブ、53は反応器の圧力を一定に制御する圧力制御バルブである。

【0067】plasma/ O_2 アニールの手順は次のとおりである。所定の温度に加熱されているサセプタ44の上に処理前の複合構造体43を載せ、真空ポンプ42を作動させると同時に、排気バルブ52を開口し、直ちに反応器41内を真空排気する。反応器41内部の圧力が 10^{-5} [Torr]台に到達したところで高純度 O_2 ガスを反応器41に所定の質量流量で導入するとともに圧力制御バルブ53を作動させて、反応器41の内圧を所定の値に調節する。内圧とサセプタ温度が安定するまで待ってから、高周波電源48の出力をオンし、サセプタ44とパワー電極45との間で放電を起こさせ、アニールを開始する。放電が起こると、 O_3 や $\text{O}(^3\text{P})$ 、 $\text{O}(^1\text{D})$ 、 $\text{O}_2(^1\Delta\text{g})$ 、などからなる活性酸素が発生する。以上の準備が整ったところで、 UV/O_2 活性酸素アニールを所定の時間、実施する。典型的なアニール条件を下に示す。

【0068】

活性酸素アニール(plasma/ O_2)条件

容器圧力 3 [mTorr]
 処理温度 400 [°C]
 サセプター-パワー電極間距離 30 [cm]
 O_2 流量 1 [cc/min]
 高周波パワー 100 [W]
 処理時間 20 [min]

所定の時間が過ぎたらアニールを終了する。すなわち、高周波出力を停止し、圧力制御バルブ53の作動を止

め、排気バルブ52を閉じ、真空ポンプ42を停止し、反応器41に空気を導入して大気開放し、複合構造体4

3を器外に取り出す。このようにして単結晶 SrTiO_3 膜と単結晶 Si 基板との複合構造体が完成する。

【0069】次に本発明の実施例3の効果を説明する。下記の表2は実施例3ならびに従来例に基づく SrTiO_3/Si 複合構造体において、 SrTiO_3 薄膜(40nm)の電界強度 $1[\text{MV}/\text{cm}]$ におけるリーク電流*(表2)

SrTiO_3/Si 複合構造体の電気特性の比較

	リーク電流 J_L [nA/cm^2]	絶縁破壊強度 BE_{ox} [MV/cm]	総電荷量 Q_{bd} [mC/cm^2]
実施例	0.089	4.9	1350
従来例	220	2.3	8.1

【0071】表2の電気特性は前記実施例と同様に MIS 容量($A1$ ゲート)に正の電圧を印加して測定した。表2から判るように、実施例3においては、従来例に比べてリーク電流 J_L を3桁強低減し、絶縁破壊強度 BE_{ox} を2倍に向上させている。さらに TDDDB 寿命に対応する総電荷量 Q_{bd} を3桁増大させ、 Q_{bd} を $1[\text{C}/\text{cm}^2]$ 台に乗せることに成功した。以上の結果は、本発明の実施例3が、結晶性酸化物薄膜/ Si 基板の複合構造体における結晶性酸化物薄膜のリーク電流や信頼性の改善に大いに寄与し得る技術であることを示している。

【0072】このような複合構造体の結晶性酸化物薄膜のリーク電流や信頼性を改善する効果は、ここまでの具体的実施例で説明した YSZ 膜や CeO_2 膜、 SrTiO_3 膜に限定されるものではなく、 Si 単結晶基板に成長させた他の単結晶酸化物薄膜や配向性酸化物薄膜にも等しく現われる効果であり、本発明の適用範囲は広い。ただし、対象とする複合構造体に応じて活性酸素アニールの最適条件は異なる。

【0073】また、結晶性酸化物薄膜の構成は説明の都合上、すべて単層膜としたが、異種の膜の積層膜でも同様の改善効果が得られる。たとえば、(001)面の CeO_2 単結晶膜と(001)面の SrTiO_3 単結晶膜を積層した $\text{CeO}_2/\text{SrTiO}_3/\text{Si}$ 複合構造体でも顕著なリーク軽減効果と絶縁破壊強度および TDDDB 耐性改善効果が現われる。

【0074】次に、本発明を電子素子に適用した実施例について説明する。

(実施例4)本実施例は、前記図1に示した IS 複合構造体を MFIS 型不揮発性メモリセルに適用した例である。

* J_L 、絶縁破壊強度 BE_{ox} および定電流 $0.1[\text{mA}/\text{cm}^2]$ を与えたときの総電荷量 Q_{bd} を比較した表である。

【0070】

【表2】

【0075】図5は MFIS 型不揮発性ランダムアクセスメモリセルの模式的構造断面図である。このセル構造は一見、前記図8で説明した従来のセル構造と同じであるが、結晶性酸化物誘電体薄膜1が活性酸素アニールを施された膜であるところが、大きく異なっている。図5において、61は p 型の(100)面の単結晶 Si 基板、62は Si 基板上に選択的に形成された厚いフィールド酸化膜で、このフィールド酸化膜62が形成されていないところがモート領域63である。メモリセルはこのモート領域63に形成されている。64はモート領域にヘテロエピタキシャル成長された単結晶酸化物誘電体膜(I)、たとえば(100)面の YSZ 膜で、素子が完成するまでの間に活性酸素アニールを施された膜である。単結晶酸化物誘電体膜64の上には強誘電体膜65、たとえば(001)面の PbTiO_3 膜とゲート電極66、たとえば Pt 電極がある。強誘電体膜65は結晶性酸化物誘電体薄膜64の上にヘテロエピタキシャル成長している。67、68は n 型不純物、たとえば P や As が高濃度にドーピングされているソース電極とドレイン電極である。

【0076】次に、この素子の製造工程の要点を簡単に説明すると、初めに p 型の(100)面の単結晶 Si 基板の表面の汚染物を RCA 洗浄で除去した後、 Si 基板前面に熱酸化法で40[nm]の熱酸化膜を形成し、さらにその上に150[nm]の Si_3N_4 膜(以下 SiN 膜と略記する)を LPCVD 法で成膜する。つづいて、ホトリソグラフィとプラズマエッチング法を用いて、モート領域に相当する部分を残して SiN 膜(および熱酸化膜)を除去する。次に、レジストを剥離し、再び RCA 洗浄した後、熱酸化法(LOCOS 酸化)で400

50[nm]のフィールド酸化膜を形成する。このとき、 S

i N膜が形成されている部分には酸化膜は成長しないのでフィールド酸化膜と一緒にモート領域も形成される。この後、熱燐酸でSi N膜を除去する。

【0077】 つづいて、モート領域に単結晶(001)面のYSZ膜の成長を行う。RCA洗浄と希弗酸洗浄とでモート領域の汚染物と自然酸化膜を除去したあと、基板全面に約13モル%の Y_2O_3 を含有するYSZ膜を20[nm]、高真空電ビーム蒸着法にて成長する。この成膜については前述の複合構造体の実施例で説明したので、ここでは説明を繰り返さない。YSZ膜の成長が終

わったあと、基板を1%濃度の希弗酸に約10秒浸漬*

*せ、超純水でリンスし乾燥する。

【0078】 希弗酸による洗浄が終わったところで、MOCVD(有機金属気相成長)法を用いて(001)面の単結晶 $PbTiO_3$ の成長を行う。使用する原料はテトラエチル鉛 $[Pb(C_2H_5)_4]$ とテトライソプロポキシチタン $[Ti(i-OC_3H_7)_4]$ と乾燥酸素(O_2)である。前二者は常温常圧で液体であり、いずれも気化器で気化し乾燥Arキャリアガスで反応器に輸送する。成長装置は特別な装置ではなく汎用機を用いる。代表的な成膜条件は下記のとおりである。

【0079】

(001)面の単結晶 $PbTiO_3$ のMOCVD成長条件

成長圧力	6[Torr]
成長温度	640[°C](基板温度)
原料	$Pb(C_2H_5)_4$, $Ti(i-OC_3H_7)_4$, O_2
気化温度	-15°C [$Pb(C_2H_5)_4$] 25°C [$Ti(i-OC_3H_7)_4$]
ガス流量	
O_2	100cc/min
Pb原料キャリア(Ar)	25cc/min
Ti原料キャリア(Ar)	22cc/min
Ar希釈	200cc/min

次に活性酸素アニール、たとえばUV/ O_3 アニールを所定の時間実施する。UV/ O_3 アニールの詳細な説明はすでに述べているので省略する。この活性酸素アニールによって、先に成長したYSZ膜ばかりではなく $PbTiO_3$ 強誘電体膜のリーク電流特性と絶縁破壊強度、TDD B耐性を同時に改善することができる。なお、活性酸素アニールをYSZ膜形成の直後に行っても、無論良いが、この場合には強誘電体膜の電気特性の改善は望め※

※ない。

【0080】 強誘電体膜の成長を終了したら、次にゲート電極材料、たとえば白金Ptの全面成膜を行う。Ptの成膜は汎用のDCマグネトロンスパッタリング法を用いる。電極材料としては99.99%純度以上のPtターゲット、膜厚は100[nm]厚である。Ptの典型的なスパッタリング成膜条件は次のとおりである。

【0081】

PtのDCマグネトロンスパッタリング条件

成膜圧力	3[mTorr]
成膜温度	400[°C](基板温度)
原料	99.99%Ptターゲット
スパッタガス	Ar
ターゲット-基板距離	30[cm]
DCパワー	150[W]

Ptの成膜が済んだら、フォトリソグラフィとArスパッタエッチング技術を用いてPtゲート電極を形成する。Ptのエッチングが終了したレジストをそのままにして続けて $PbTiO_3$ 膜とYSZ膜の反応性イオンエッチングを行う。エッチングガスは CF_4 と O_2 の混合ガスである。YSZ膜のエッチングが終了したら、レジストを灰化させて除き、基板表面の塵埃を有機溶剤を用いて超音波洗浄で除去する。

【0082】 つづいて、基板全面にPをイオン注入す

る。Si基板に注入されるのはモート領域のゲート電極を除く部分、すなわちソース、ドレイン領域である。加速電圧は100[keV]、ドーズ量は 5×10^{15} [個/ cm^2]である。イオン注入が終了したら、ソース、ドレインの注入不純物の活性化を急速熱処理(RTA)装置を用いて行う。熱処理温度は1200[°C]、熱処理時間は30秒である。このようにしてMFIS型の強誘電体メモリセルが完成する。

【0083】 (実施例5) 電子素子の他の実施例は、前

記本発明の I S 複合構造体を用いた S O I 基板上に MOS トランジスタを形成した例である。図 6 は S O I 構造の MOS トランジスタの模式的構造断面図である。この構造は図 9 で説明した従来例の構造と見かけ上、同じであるが S i エピタキシャル膜と S i 単結晶基板との間に挟持された結晶性酸化物誘電体薄膜が活性酸素アニールを施された膜であるところが、大きく異なっている。

【0084】図 6 において、71 は高濃度に不純物ドーブされた n 形の (100) 面の単結晶 S i 基板、72 は S i 基板 71 上にヘテロエピタキシャル成長された単結晶酸化物誘電体膜 (I)、たとえば (100) 面の Y S Z 膜で、素子が完成するまでの間に活性酸素アニールを施された膜である。73 は Y S Z 膜 72 の上にヘテロエピタキシャル成長した n 形または p 形の比較的厚い (100) 面の単結晶 S i 膜である。74 は単結晶 S i 膜 73 上に形成された厚いフィールド酸化膜、75 はモート領域である。モート領域 75 内にある 76 はポリシリコンゲート電極であり、これと単結晶 S i 膜 73 との間には S i O₂ 熱酸化膜 (ゲート酸化膜) 77 が形成されている。また 78、79 は p 形または n 形の不純物が高濃度にドーブされているソース電極とドレイン電極である。

【0085】製造工程の要点を簡単に説明すると、初めに、高濃度にドーブされた n 形の (100) 面の単結晶 S i 基板 61 を R C A 洗浄と稀弗酸洗浄とで表面を清浄にし、その全面に (001) 面の Y S Z 膜を 150 [nm]、すでに説明した高真空電極蒸着法にて成長させる。成長後、この I S 複合構造体に、前に説明した活性酸素アニール、例えば、p l a s m a / O₂ アニールをやや長めに施す。このあと、基板を 1% 濃度の稀弗酸に約 10 秒浸漬させ、超純水でリンスし乾燥する。つづいて (100) 面の Y S Z 膜 72 の表面に 2 μm の厚みの (100) 面の単結晶 S i を周知の常圧 C V D 法でエピタキシャル成長させる。原料は S i C l₄ と H₂ である。このようにして低リーク電流、高絶縁破壊耐性の S O I 基板ができる。

【0086】次にエピタキシャル S i 膜全面に p 形の不純物 B を 3×10^{13} [個/cm²] イオン注入し、基板表面を R C A 洗浄で清浄にする。前出の実施例とまったく同じ L O C O S 酸化法でフィールド酸化膜 74 とモート領域 75 を形成する。L O C O S 酸化工程を経るあいだに先にイオン注入した B 不純物はエピタキシャル S i 膜内に広く拡散しかつ活性化する。

【0087】つづいて、基板を熱酸化してモート領域にゲート酸化膜となる薄い S i O₂ 熱酸化膜 77 (厚み 20 [nm]) を成長させ、さらにこの上に L P C V D 法で膜厚 300 [nm] のポリシリコン膜を堆積した後、フォトリソグラフィと反応性イオンエッチングを用いて、図 6 に示すように、ポリシリコンゲート電極 76 を形成する。

【0088】レジストを除去した後、基板表面に P をイオン注入する。この注入でエピタキシャル S i 膜に B が打ち込まれるのはモート領域のゲート電極を除く部分、すなわちソース、ドレイン領域である。ポリシリコンゲート電極にも B が同時に打ち込まれる。加速電圧は 100 [keV]、ドーズ量は 5×10^{15} [個/cm²] である。

【0089】イオン注入が終了したら、基板表面を R C A 洗浄で清浄にした後、ソース電極 78、ドレイン電極 79、ゲート電極 76 の不純物を活性化するために、1120 [°C] に加熱した拡散炉内で 2 時間のドライブインを行う。このようにして本発明に基づく、S O I 基板を用いた MOS トランジスタが完成する。

【図面の簡単な説明】

【図 1】本発明に基づく複合構造体の基本構造を示す断面図。

【図 2】本発明の製造方法で使用する活性酸素アニール装置の一例を示す断面図。

【図 3】本発明の製造方法で使用する活性酸素アニール装置の他の一例を示す断面図。

【図 4】本発明の製造方法で使用する活性酸素アニール装置の他の一例を示す断面図。

【図 5】本発明の電子素子の一例として不揮発性メモリセルの構造を示す断面図。

【図 6】本発明の電子素子の一例として S O I 基板上に MOS トランジスタを形成した構造を示す断面図。

【図 7】結晶性酸化物誘電体薄膜のリーク電流特性を示す特性図。

【図 8】従来の電子素子構造に基づく不揮発性メモリセルの一例の断面図。

【図 9】従来の電子素子構造に基づく MOS トランジスタの一例の断面図。

【符号の説明】

- | | |
|---|--------------|
| 1…単結晶 S i 基板 | 2…結晶性酸化物誘電体膜 |
| 11…反応器 | 12…真空ポンプ |
| 13…アニール前の Y S Z / S i 複合構造体 | |
| 14…サセプタ | 15…酸素ボンベ |
| 16…整圧器 | 17…質量流量計 |
| 18…誘電体バリア放電エキシマランプ | |
| 19…合成石英窓 | 20…排気バルブ |
| 21…大気排出バルブ | 22…真空排気装置 |
| 31…アニール前 C e O ₂ / S i 複合構造体 | |
| 32…オゾン発生器 | 33…低圧水銀ランプ |
| 41…反応器 | 42…真空ポンプ |
| 43…アニール前 S r T i O ₃ / S i 複合構造体 | |
| 44…サセプタ | 45…パワー電極 |
| 46…ブロッキング容量 | 47…整合器 |

48…高周波電源
50…整流器
52…排気バルブ
61…Si基板
酸化膜
63…モート領域
体薄膜
65…強誘電体膜

49…酸素ポンプ
51…質量流量調
53…圧力制御バ
62…フィールド
64…単結晶誘電
66…ゲート電極

67…ソース電極
71…Si基板
体薄膜
73…単結晶Si膜
酸化膜
75…モート領域
ンゲート電極
77…ゲート酸化膜
79…ドレイン電極

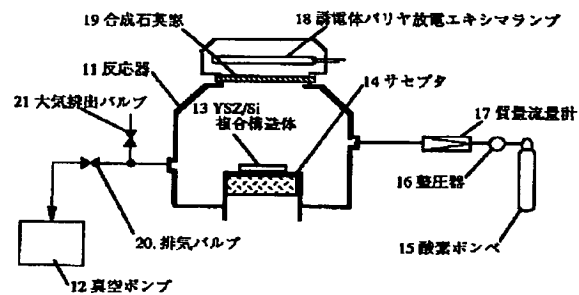
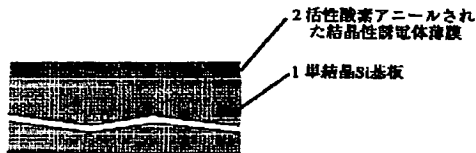
68…ドレイン電
72…単結晶誘電
74…フィールド
76…ポリシリコ
78…ソース電極

【図1】

【図2】

(図1)

(図2)

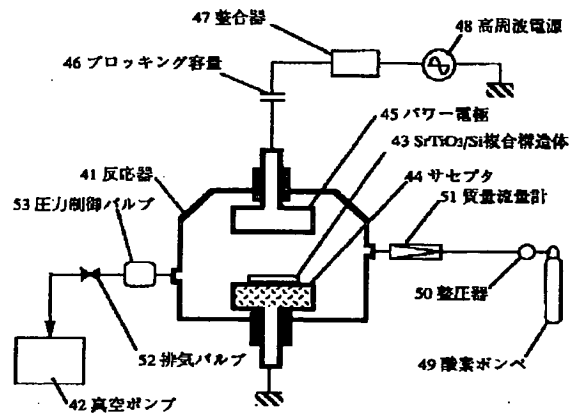
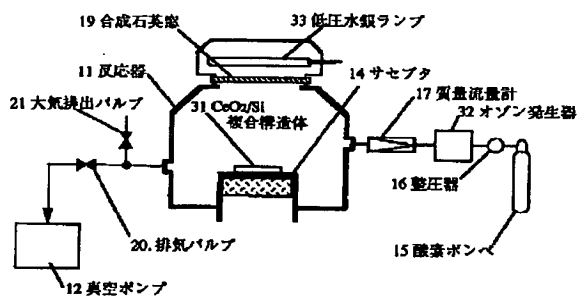


【図3】

【図4】

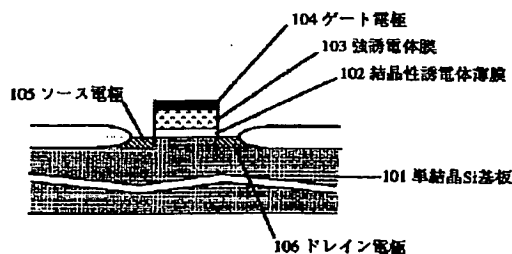
(図3)

(図4)



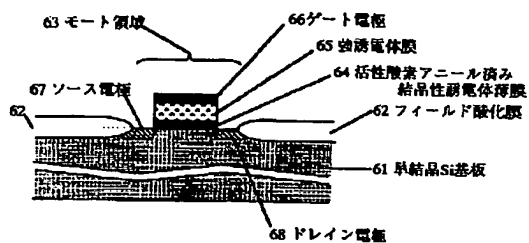
【図8】

(図8)



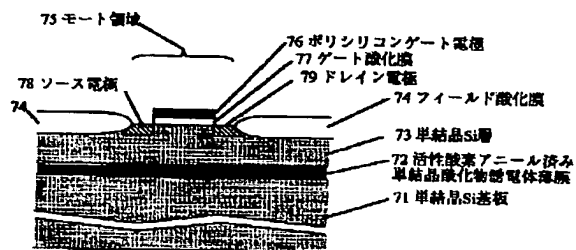
【図5】

(図5)



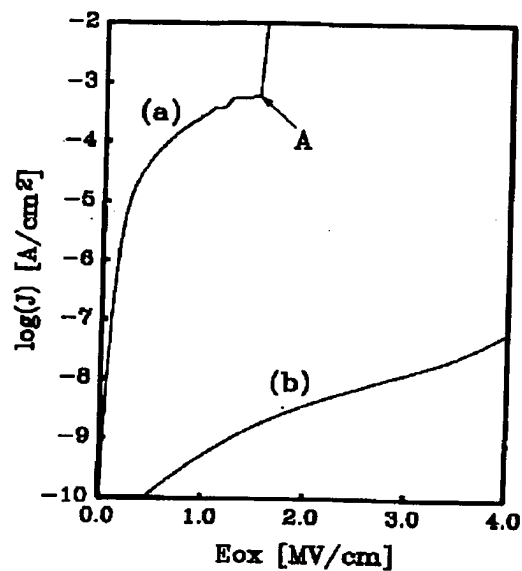
【図6】

(図6)



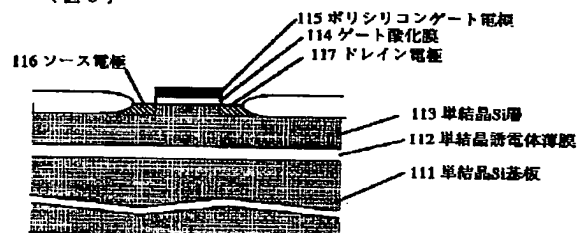
【図7】

(図7)



【図9】

(図9)



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* NOTICES *

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1. This document has been translated by computer. So the translation may not reflect the original precisely.
2. **** shows the word which can not be translated.
3. In the drawings, any words are not translated.

CLAIMS

[Claim(s)]

[Claim 1] The composite-construction object of the crystalline oxide dielectric thin film and single-crystal-silicon base which are characterized by the aforementioned crystalline oxide dielectric thin film being a thin film by which it was heat-treated in active oxygen atmosphere after growth in the composite-construction object which grew up the crystalline oxide dielectric thin film into the single-crystal-silicon semiconductor base.

[Claim 2] The electronic device to which the aforementioned composite-construction object is characterized by being the thin film by which it was heat-treated once [at least] in active oxygen atmosphere by the time it completed electronic device structure after the formation in the electronic device which includes the composite-construction object into which the crystalline oxide dielectric thin film was grown up in a single-crystal-silicon semiconductor base.

[Claim 3] The composite-construction object according to claim 1 with which the aforementioned crystalline oxide dielectric thin film is characterized by being the single crystal film which carried out hetero-epitaxial growth at the aforementioned single-crystal-silicon base, or an electronic device according to claim 2.

[Claim 4] A composite-construction object or an electronic device given in any of the claim 1 characterized by the bird clapper, or a claim 3 they are from the multilayer in which the aforementioned crystalline oxide dielectric thin film was formed of the laminating of two or more material.

[Claim 5] A composite-construction object or an electronic device given in any of the claim 1 characterized by being the monolayer or cascade screen from which the aforementioned crystalline oxide dielectric thin film was constituted by one or more material chosen from a stabilized zirconia, the cerium oxide, the strontium titanate, the magnesium oxide, and the yttrium oxide, or a claim 4 they are.

[Claim 6] The above-mentioned electronic device carries out the laminating of a crystalline oxide dielectric thin film (I), a ferroelectric thin film (F), and the metal-electrode film (M) to a part of single-crystal-silicon semiconductor base (S) at order. It is the MFIS transistor of the agensis section of this cascade screen which prepared the source electrode and the drain electrode in part at least. An electronic device given in any of the claim 2 characterized by this crystalline oxide thin film being a thin film by which it was heat-treated once [at least] in active oxygen atmosphere after growth or ferroelectric thin film growth, or a claim 5 they are.

[Claim 7] The above-mentioned electronic device is an electronic device given in any of the claim 2 characterized by being the SOI substrate which grew epitaxially one or more crystalline oxide dielectric thin films (I) and single-crystal-silicon films on the single-crystal-silicon semiconductor base (S), and this crystalline oxide thin film being a thin film by which it was heat-treated once [at least] in active oxygen atmosphere after the growth, or a claim 5 they are.

[Claim 8] How to heat-treat under the UV irradiation in which heat treatment in the aforementioned active oxygen atmosphere contains the wavelength of less than 190nm while supplying oxygen, The method of heat-treating, while supplying ozone, the method of heat-treating under oxygen plasma, A composite-construction object given in any of the claim 1 characterized by being attained by the method

of overlapping when it differs, and performing the method of any one **, the method of performing simultaneously combining two or more methods, or two methods or more, or a claim 7 they are, or the manufacture method of an electronic device.

[Claim 9] A composite-construction object given in any of the claim 1 characterized by carrying out heat treatment in the aforementioned active oxygen atmosphere immediately after crystalline oxide dielectric thin film formation, or a claim 8 they are, or the manufacture method of an electronic device.

[Claim 10] The manufacture method of an electronic device given in any of the claim 2 characterized by being given after heat treatment in the aforementioned active oxygen atmosphere forms a crystalline oxide dielectric thin film and forms at least one of a conductive oxide or an oxide-semiconductor film, an oxides-superconductors film, and the oxide ferroelectric films on this crystalline oxide dielectric thin film, or a claim 8 they are.

[Translation done.]

* NOTICES *

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2. **** shows the word which can not be translated.
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DETAILED DESCRIPTION

[Detailed Description of the Invention]

[0001]

[The technical field to which invention belongs] this invention relates to the manufacture method of these structures or an electronic device at the composite-construction object of the very few crystalline oxide dielectric thin film of a leakage current, and a single-crystal-silicon (Si) substrate, the structure of the electronic device using this, and a row. In addition, the crystalline dielectric thin film said here shall mean the dielectric thin film which carried out priority orientation in the single crystal or the specific direction.

[0002]

[Description of the Prior Art] Generally the single crystal film has the feature that many physical properties flat in crystal structure and peculiar to the material, such as eye a uniform hatchet, conductivity, a dielectric, insulation, a ferroelectricity, superconductivity nature, a semiconductor property, light-transmission nature, optical magnetic properties, and crystal periodicity, appear keenly strongly, compared with the amorphous film or the polycrystal film. Therefore, it is useful in various industrial fields, and the realization is desired strongly. In recent years, these single crystal film is loaded into Si semiconductor device or an integrated circuit, and the attempt which is going to realize the high integrated circuit and high smart device of the added value which cannot be attained only with the conventional Si system material is performed actively, and is come and required. Basic structure indispensable when such a new Si semiconductor device is constituted is the composite-construction object (the following information-separator structure and abbreviated name) of a crystalline dielectric thin film (I) and a single crystal Si (S).

[0003] When an example is given, there is a MFIS type nonvolatile random access memory cell as shown in the typical structure section view of drawing 8 . This cell replaces O portion of an MOS transistor which consists of a (Metal M)-silicon-oxide (O)-semiconductor (S) by the single crystal ferroelectric film (F) which has electrically the spontaneous polarization which can be reversed, and the single crystal dielectric film (I).

[0004] In drawing 8 , it is the single crystal oxide dielectric film (I) by which 101 was carried out at the single crystal Si substrate of a p type field (100), and hetero-epitaxial growth of 102 was carried out on the Si substrate 101, for example, (100) is the YSZ film of a field. 103 is the ferroelectric film by which hetero-epitaxial growth was carried out on the single crystal oxide dielectric film 102, for example, (001) is PbTiO₃ film of a field. 104 is the gate electrode formed on the ferroelectric film 103, for example, is Pt electrode. 105 and 106 are n type source electrodes and drain electrodes, they carry out the ion implantation of P (Lynn) or the As (arsenic), respectively, make it activated by thermal diffusion, and are formed.

[0005] By this memory, information is recorded by impressing a positive or negative pulse voltage between gate (electrode G)-Si substrates (Sub), carrying out reversal fixation of the spontaneous-polarization vector of a ferroelectric film, and making a transistor into a flow or non-switch-on.

[0006] Moreover, as the above-mentioned ferroelectric layer, PbTiO₃ (lead titanate) film formed by the

chemical vapor growth (CVD), the magnetron sputtering method, etc., $\text{Pb}(\text{Zr}_x, \text{Ti}_{1-x})\text{O}_3$ (zircon lead titanate) film, $3\text{OBi}_4\text{Ti}_{12}$ (titanic-acid bismuth) film, etc. are examined (all are polarization shaft orientation single crystal films).

[0007] Moreover, as the above-mentioned single crystal dielectric layer, CeO_2 (cerium oxide) film formed by the electron-beam-evaporation method or CVD, a YSZ (ITTORIYA stabilization JIRUKONIYA) film, SrTiO_3 (strontium titanate) film, etc. are examined. These single crystal dielectric films are bearing a role of a buffer coat which prevents that function as a template layer for forming a single crystal ferroelectric film, and also Si substrate carries out counter diffusion to a ferroelectric.

[0008] Moreover, MFMIS structure is also proposed by the nonvolatile random access memory cell besides the above-mentioned MFIS structure. Single crystal F layer and M layers inserted between I layers of single crystals are here, the electric conduction films, for example, Pt film etc., of a single crystal etc. information-separator structure is used for the lower part of MFM structure also in this case.

[0009] Next, the MOS transistor of the SOI structure shown in the typical structure section view of drawing 9 is also the important application of information-separator structure. The MOS transistor of this structure has the feature which should be mentioned especially that it can escape from a very detrimental latch up completely on operation.

[0010] In drawing 9, they are single crystal dielectric films, such as CeO_2 by which hetero-epitaxial growth was carried out by 111 touching Si substrate of a field (100) and 112 touching this Si substrate 111, YSZ, and SrTiO_3 . This Si substrate 111 and the single crystal dielectric film 112 have constituted information-separator structure. The single crystal Si layer of the p type field (100) where hetero-epitaxial growth of 113 was carried out by CVD on the single crystal dielectric film 112, the SiO_2 gate oxide film in which 114 was formed by thermal oxidation of the single crystal Si layer 113, and 115 are the gate electrodes of contest polysilicon formed by CVD and dry etching on the gate oxide film 114. 116 and 117 are the source electrodes and drain electrodes of N type, they carry out the ion implantation of P (Lynn) or the As (arsenic), respectively, make it activated by thermal diffusion, and are formed.

[0011] Moreover, the oxide high-temperature superconductivity thin film wiring tried aiming at high speed processing of an integrated circuit also forms quality single crystal superconductor film (for example, $\text{YBa}_2\text{Cu}_3\text{O}_7$) wiring on information-separator structure. As I layers, YSZ and the single crystal cascade screen of Y_2O_3 are used, for example.

[0012] Thus, it is understood that information-separator structure is the important basic structure for realizing various devices containing the functional thin film (a ferroelectric film, a semiconductor film, an electric conduction film, a superconductivity film, dielectric film) of a single crystal on a single crystal Si substrate. However, in the present condition, the problem that a low and (3) passage-of-time destructive (TDD) life have short (2) disruptive strengths with large (1) leakage current is in I layers of single crystal oxide dielectric thin films, and it has been a serious obstacle at realization of the above highly efficient devices.

[0013] The above-mentioned problem is explained in detail based on actual data. Drawing 7 is the leakage-current property view of a crystalline oxide dielectric thin film. It is that a characteristic curve (a) indicates the example of a property of the conventional single crystal YSZ film to be in drawing 7. On the YSZ film of the field (100) which grew epitaxially Si substrate of an N type single crystal (100) side Leakage-current density J [A/cm^2] which impressed and measured positive voltage to the gate electrode of the MIS capacity which formed and produced aluminum electrode (gate) of a diameter 200 [μm] is shown as a function of field strength E_{ox} [V/cm]. In addition, in order to remove an absorption current component, measurement of current is performed, after carrying out voltage impression and time passes enough. Moreover, thickness is 40 [nm] and the forming-membranes method is an electron-beam-evaporation method here. The forming-membranes methods (vacuum evaporation conditions etc.) are later mentioned in the example of this invention in the cleaning-method row of a substrate.

[0014] The leakage current of the single crystal YSZ film of the conventional example is in a very high level so that the characteristic curve (a) of drawing 7 may be seen and may be known. Considering practical use, I want to suppress leakage-current density below to $J = 1$ [nA/cm^2] at the time of field strength $E_{\text{ox}} = 1$ [MV/cm]. However, in the above-mentioned conventional example, this and the value

which was greatly different widely are indicated to be the bases of $J = 100$ [$\mu\text{A}/\text{cm}^2$]. Moreover, the current jump of A points in the characteristic curve (a) of drawing 7 shows that the YSZ film is carrying out dielectric breakdown on this voltage ($E_{ox} \approx 1.5$ [MV/cm]). In order to apply to the above-mentioned MFIS nonvolatile memory cell etc., it is the value which cannot be said that this insulating disruptive strength is enough.

[0015] It is indispensable conditions that there is long-term dielectric-breakdown (TDDB) resistance over power supply stress in I more layers. This resistance measures and evaluates the amount Qbd of net charge [C/cm^2] which passed the film by the time it generally gave and carried out dielectric breakdown of the constant-current stress to the dielectric film. The typical TDDB life when giving the constant-current stress of $J = 0.1$ [mA/cm^2] to the above-mentioned single crystal YSZ film was $Q_{bd} = 1.2$ [mC/cm^2], respectively. This value is a low comparable, if it compares with typical value $Q_{bd} = 10$ [C/cm^2] of thermal oxidation SiO_2 film used as gate oxide films, such as an MOS transistor. In order for power supply stress to apply a YSZ film to a comparatively strong MFIS nonvolatile memory cell etc., the TDDB life of the base of at least 0.1 [C/cm^2] is required.

[0016] Such a problem of I layers in information-separator structure is not having restricted to the single crystal YSZ film formed by the electron-beam-evaporation method, but is a problem by which the YSZ film produced by other single crystal oxide dielectric-materials film and other forming-membranes methods is also observed in common.

[0017]

[Problem(s) to be Solved by the Invention] It is made in order that this invention may solve the trouble of the crystalline dielectric film (I layers) of the device which used such the conventional information-separator structure and this, and there are few leakage currents, dielectric-breakdown pressure-proofing is high, and the with-time dielectric-breakdown life aims at realizing a composite-construction object, the electronic devices using it, and those manufacture methods with information-separator structure, i.e., a crystalline oxide dielectric thin film, of having a crystalline, sufficiently long oxide dielectric layer, and a single-crystal-silicon base.

[0018]

[Means for Solving the Problem] In order to attain the above-mentioned purpose, as information-separator composite-construction object, the electronic devices using this, and those manufacture methods were indicated to each patent claim, it constituted in this invention. That is, in invention according to claim 1, I layers presupposed that it is the crystalline oxide dielectric thin film which heat-treated in active oxygen atmosphere after growth in information-separator composite-construction object which grew up the crystalline oxide dielectric thin film into the single-crystal-silicon semiconductor base.

[0019] In the electronic device built on this in the claim 2 including information-separator composite-construction object, I layers presupposed similarly that it is the crystalline oxide dielectric thin film by which it was heat-treated in active oxygen atmosphere by the time it completed the electronic device after growth (active oxygen annealing is called below).

[0020] Claims 3-5 show the concrete mode of a claim 1 or a claim 2. That is, in a claim 3, it constitutes from a single crystal film which made the single-crystal-silicon base carry out hetero-epitaxial growth of the crystalline oxide dielectric thin film. Moreover, in a claim 4, the multilayer formed of the laminating of two or more oxide dielectric films constitutes the aforementioned crystalline oxide dielectric thin film. Moreover, it is supposed that it is the monolayer or laminating bipolar membrane as which the crystalline oxide dielectric thin film was chosen from a stabilized zirconia, the cerium oxide, the strontium titanate, the magnesium oxide, and the yttrium oxide in the claim 5.

[0021] Moreover, a claim 6 and a claim 7 show the concrete mode of an electronic device, and a claim 6 shows the case where the electronic device of a claim 7 is a SOI (Silicon On Insulator) substrate, when an electronic device is a MFIS transistor (MFIS type nonvolatile random access memory cell).

[0022] A claim 8 is a composite-construction object given in the aforementioned claim, or the manufacture method of an electronic device, and is in a manufacturing process. furthermore, active oxygen annealing of a crystalline oxide dielectric thin film How to heat-treat under the UV irradiation

which contains the wavelength of less than 190nm while supplying oxygen, The method of heat-treating, while supplying ozone, the method of heat-treating under oxygen plasma, The method of any one * , or the method of performing simultaneously combining two or more methods (for example, it heat-treats, supplying ozone under UV irradiation), Or it is attained by the method (the following method is performed after performing one method) of overlapping when it differs, and performing two or more methods. In addition, the range which has the capacity which decomposes oxygen and is made into active oxygen is meant in the wavelength of less than 190nm of the above-mentioned ultraviolet rays.

[0023] Moreover, a claim 9 is one mode of the manufacture method, and heat treatment in active oxygen atmosphere is carried out immediately after crystalline oxide dielectric thin film formation. Moreover, a claim 10 is one mode of the manufacture method of the aforementioned electronic device, and after heat treatment in active oxygen atmosphere forms a crystalline oxide dielectric thin film and forms at least one of a conductive oxide or an oxide-semiconductor film, an oxides-superconductors film, and the oxide ferroelectric films on this crystalline oxide dielectric thin film, it is given.

[0024] Hereafter, an operation is explained. First, the mechanism of the leakage current of the crystalline oxide thin film formed on the single crystal Si substrate is explained. According to the place which this invention person studied wholeheartedly and was clarified, the metal Si element of the oxygen vacancy inside the crystal grain of (1) oxide, the metallic element void inside (2) crystal grain, the energy level of (3) grain-boundary side, and (4) crystal-grain gap etc. is one of the causes of the leakage current of a crystalline oxide thin film, and a different (separation is impossible) leakage current based on each cause flows. The above-mentioned cause (1) and (2) are the so-called point defects from which the atom which should be in the lattice point of a crystal escapes probable. If a metallic element falls out, it will escape also from the oxygen atom combined with this in many cases together, and it will serve as a metal-oxygen void. An electron can move now easily between level and these point defects will cause the so-called hopping conduction, if electronic level is formed in the forbidden band of an oxide and this density becomes more than fixed. An amorphous film is also often observed and such a point defect causes a leakage current too.

[0025] The above-mentioned cause (1) and (2) are the causes peculiar to a crystalline oxide film about the leakage current to which a cause (3) and (4) flow a grain-boundary side to having been the cause of the leakage current which flows the inside of a grain boundary. Although an impression with strange "grain boundary" may be received in the crystalline oxide thin film which is growing epitaxially to the single crystal Si here, when this makes a single crystal Si carry out hetero-epitaxial growth of the dissimilar material from which a lattice constant is delicately different, in order to open the stacking-fault stress of a hetero side, it is the grain boundary produced inevitably, and the grain-boundary side is perpendicularly prolonged in the growth side. In this grain-boundary side, a crystal period is interrupted and many things which have not carried out a chemical bond exist in the atom which exists in a grain-boundary side. The above-mentioned cause (3) is the electronic level formed in the interior of a crystalline oxide forbidden band as a result in which these uncombined atoms exist. If electronic level becomes high concentration, a band will be formed and a leakage current will be increased. A cause (4) is the metal Si atom which invaded from Si substrate at the time of membrane formation of a crystalline oxide thin film, and segregated to the grain boundary. Since there is an opening about a 1 - number atomic layer in the grain boundary, the trap of the Si which has invaded from the substrate at the time of growth under reduced pressure (under hypoxia concentration) is carried out to a grain-boundary gap with a metal state (or low valence state). Naturally Si segregated to the grain boundary becomes the electric conduction path of a leakage current.

[0026] or this invention solves the conventional problem containing a leakage current how in the place which finished explaining the mechanism of generating of a leakage current (operation) -- ***** -- it explains According to this invention person's latest research result, it is thought that reduction of the leakage current by this invention takes place based on the following mechanisms. At active oxygen annealing of this invention, active oxygen, such as reactant high O (1D), and O₂ (1deltag), O (3P), heat-treats comparatively a crystalline oxide thin film / single crystal Si structure at low temperature by the

atmosphere abundantly generated in the gaseous phase. At this time, active oxygen is diffused the inner direction toward Si substrate. A part of active oxygen encounters the oxygen vacancy of a cause (1), it extinguishes a vacancy (restoration), and is stabilized. Moreover, toward a grain boundary, a part of other active oxygen is diffused along with a grain boundary, and it oxidizes, sets to SiO₂ the metal-like silicon which is carrying out the segregation to the grain boundary, and fills the grain-boundary gap section in SiO₂ network. The metal Si of a cause (4) is removed by this. In addition, the above-mentioned heat treatment temperature is made into the range from ordinary temperature to about about 500 degrees C, and its about 200-500 degrees C are especially desirable. The reason is because active oxygen changes to usual oxygen at the elevated temperature of 500 degrees C or more and annealing time becomes long below 200 degrees C.

[0027] Furthermore, the uncombined hand atom M which existed in the grain boundary side is process in which SiO₂ network is formed in a grain-boundary gap, accomplishes approaching Si and M-O-Si combination, and cancels the uncombined hand. A cause (3) is canceled by this. The remaining active oxygen is diffused to a single crystal Si substrate, and grows up a very thin SiO₂ film into a crystalline oxide thin film / Si interface. Although growth of SiO₂ is generally accompanied by expansion of volume, since it is difficult to overcome compressive stress like the usual thermal oxidation, and to grow up, by active oxygen annealing, some Si elements are emitted into an oxide film, and oxidization by the interface advances at active oxygen annealing whose heat treatment temperature is low temperature, pressing down expansion of volume. Si atom emitted here encounters a metal-oxygen void in the process which carries out an out diffusion, and is captured in a metal atom position. On the other hand, at this time, from the gaseous phase, active oxygen is spread the inner direction, it enters in the middle of two or more metals of an oxide and Si atom which it is possible to encounter Si atom captured by the void, and approach a void, and M-O-Si combination is completed. In this way, the metal-oxygen void of a cause (2) is filled up by the Si-Ox (x=4-5) atomic group, and disappears. In addition, a large majority of Si atoms which were not captured by the void deposit on the front face of an oxide thin film. Since this sludge checks epitaxial growth of other thin films to a crystalline oxide thin film top, in this invention, it is removed by the fluoric acid solution etc. before growth.

[0028] In this way, according to the method, above-mentioned cause [of a leakage current] (1) - (4) is altogether removable in the structure row of this invention. Consequently, this invention can reduce the leakage current of a crystalline oxide thin film on the level of several figures compared with the conventional example so that it may mention later in the example.

[0029] An improvement of the amount Qbd of net charge which shows the insulating disruptive strength BEox and TDDb life by this invention below is described. It is hard to say that dielectric breakdown of crystalline oxide thin films, such as CeO₂ made into the object of this invention and YSZ, and the mechanism of fatigue are in a research stage, and are fully solved. However, between field strength Eox, the amount Qbd of net charge, and leakage-current J, it is admitted widely that there is strong positive correlation. This has suggested that the factor and the cause of a leakage current of determining Eox and Qbd are common. Since it can remove the cause of the leakage current of a crystalline oxide as mentioned above, this invention can remove this, dielectric breakdown concerned deeply, and the factor of fatigue, as a result, compared with the conventional example, is high in the insulating disruptive strength BEox, and can enlarge the amount Qbd of net charge.

[0030]

[Effect of the Invention] As explained above, in this invention, it writes in the composite-construction object of a crystalline oxide dielectric thin film and a single crystal Si substrate as the film which gave active oxygen annealing after growing up this crystalline dielectric thin film, and the effect that (1) leakage-current property in a crystalline dielectric film, (2) insulation disruptive strength, and (3) with-time dielectric-breakdown (TDDb) resistance are improvable is acquired.

[0031] Moreover, in this invention, the effect that the destruction and the malfunction of an electronic device which originated in the poor property of above-mentioned (1) - (3), and had been produced are cancelable is acquired by applying the composite-construction object by the above-mentioned this invention to an electronic device including the composite-construction object of a crystalline oxide

dielectric film and a single crystal Si substrate.

[0032]

[Embodiments of the Invention] The manufacture method of the electronic device which includes the composite-construction object of the crystalline oxide dielectric thin film and single crystal Si substrate concerning this invention and this composite-construction object hereafter, and a composite-construction object and an electronic device is explained based on the gestalt and example of operation.

[0033] (Gestalt of operation) Drawing 1 is the cross section showing the gestalt of 1 operation of the composite-construction object of the crystalline oxide dielectric thin film based on this invention, and a single crystal Si substrate. The single crystal Si substrate in which it sets to drawing 1 and 1 has the predetermined crystal faces, such as (100), (111), and a field (110), and 2 are the crystalline oxide dielectric thin film by which hetero-epitaxial growth was carried out on the Si substrate 1 in chemical vapor growths, such as physical vapor growths, such as an electron-beam-evaporation method, the ion beam sputtering method, a molecular beam epitaxy method, and the laser ablation method, and CVD, or a single crystal oxide dielectric thin film. Although the zirconia film (for example, YSZ film) stabilized by Y₂O₃, Sm₂O₃, CaO, etc., a cerium-oxide (CeO₂) film, a strontium-titanate (SrTiO₃) film, a yttrium-oxide (Y₂O₃) film, a magnesium oxide (MgO), etc. are mentioned as a dielectric thin film applicable to this, if it is the oxide dielectric thin film by which is not limited to this and hetero-epitaxial growth is carried out at Si substrate, other thin films are sufficient. After the oxide dielectric thin film 2 grows up to be the Si substrate 1, a predetermined time and heat treatment are performed to it at the temperature of under 500 [°C] from ordinary temperature in active oxygen atmosphere, such as the Si substrate 1O₃, O (1P), O (1D) and O₂ (1delta), and O₂~. In this way, the composite-construction object based on this invention is completed.

[0034] Below, the example of the manufacture method of this invention will also be collectively explained by showing the process which produces this, while showing the concrete example of the form of operation of the above-mentioned this invention composite-construction object.

[0035] (Example 1) The concrete example of the beginning of the composite-construction object based on this invention is an example which builds the composite-construction object which consists of a single crystal Si substrate 1 of a field (100), and a single crystal ITTORIYA stabilized-zirconia (YSZ) thin film 2 of the epitaxially grown field (100) using a high-vacuum electron-beam-evaporation method and UV/O₂ active-oxygen annealing. UV means the ultraviolet rays as a generation means of active oxygen here, and O₂ means the oxygen as a raw material of active oxygen.

[0036] First, the surface contamination and surface natural oxidation object of the single crystal Si substrate 1 of a field (100) are removed by RCA washing (traditional Si substrate cleaning method which consists of washing by the mixed liquor of aqueous ammonia and hydrogen peroxide solution, and washing by the mixed liquor of a hydrochloric acid and hydrogen peroxide solution), and rare fluorine acid washing (cleaning method which carries out a rinse and which is dried with pure water after being immersed in HF solution of concentration for several 10 seconds 5%).

[0037] The YSZ film which continues and contains Y₂O₃ [about 13 mol %] on the Si substrate 1 is formed by the high-vacuum electron-beam-evaporation method. A vacuum evaporation machine is general-purpose high-vacuum type equipment marketed as the object for an experiment, or an object for mass production. The target of vacuum evaporation is the YSZ tablet with a diameter [of 15mm], and a thickness of 7mm which carried out cutting about a single crystal YSZ (Y₂O₃13-mol%) ingot.

[0038] If the procedure of membrane formation is explained, an evaporation source will be filled up with a YSZ tablet, the washed Si substrate 1 will be installed in the substrate electrode holder of a vacuum evaporation tub, and the inside of a vacuum evaporation tub will be exhausted promptly. The substrate electrode holder has the capacity to heat a substrate to the temperature from ordinary temperature to 1000 [°C], separated the distance of 30 [cm] and has opposed the source of vacuum evaporation. Exhaust air of vacuum evaporation equipment is powerful, and an internal pressure can be decompressed below to 10⁻⁹ [Torr]. If the pressure of a vacuum evaporation tub amounts to 10⁻⁶ [Torr], substrate temperature will be heated and it will be made predetermined membrane formation temperature. A high-speed electron beam is made bombardment [in the place where temperature was

stabilized at and the pressure in a vacuum evaporator tub fell below in 10^{-9} [Torr] / a tablet], and a YSZ film is grown up into Si substrate by the growth rate of 0.2 [nm/min]. Growth is continued until it gathers a growth rate to 5 [nm/min] and becomes predetermined thickness, while introducing high grade dryness oxygen into a vacuum evaporator tub from the exterior in the place where the thickness of YSZ became 3 [nm] in general and adjusting a pressure to 2×10^{-6} [Torr]. A growth rate controls and adjusts the emission current of hot cathode. (100) The typical membrane formation conditions from which the single crystal YSZ thin film of a field is obtained are as follows.

[0039]

----- YSZ membrane formation conditions Membrane formation pressure
 1×10^{-8} [Torr] (up to 3nm of thickness)

2×10^{-6} [Torr] (from 3nm of thickness to O₂ introduction)

Membrane formation temperature 800 [°C] (substrate temperature)

Target Y₂O₃:ZrO₂ = 13:87 Substrate distance 30cm Electron beam acceleration voltage The output of a vacuum evaporator power supply is cut in the place where 8kV -----

thickness became a desired value, and membrane formation is finished. Then, a substrate is annealed putting a substrate on a vacuum evaporator tub, and after substrate temperature becomes sufficiently low, a substrate is taken out from a vacuum evaporator tub. Thus, the deposited YSZ film was a single crystal epitaxial film which has fluorite structure (cubic).

[0040] The composite-construction object of the YSZ thin film and single crystal Si which were explained above as a conventional example is completed at the process so far. However, with the composite-construction object based on this invention, an active oxygen annealing process is carried out after this. Although one of the active oxygen annealing and the example of UV/O₂ annealing explain here, other below-mentioned active oxygen annealing is sufficient.

[0041] Drawing 2 is the typical important section cross section of the equipment used for annealing. In drawing 2, the reactor with which, as for 11, water cooling of the side attachment wall is carried out, and 12 are the vacuum pumps for exhausting a reactor 11, and can maintain the pressure of a reactor 11 from an ordinary pressure before 10^{-6} [Torr]. 13 is a composite-construction object in front of annealing which grew up the single crystal YSZ thin film to be a single crystal Si substrate. 14 is a susceptor for supporting 13 and holding to predetermined temperature. The composite-construction object 13 is placed so that a YSZ film surface may be upwards suitable on a susceptor 14. 15 is the chemical cylinder filled up with oxygen O₂ of 99.999% or more of purity. After O₂ of a bomb is decompressed by ready **** 16 on the occasion of annealing by the pressure about 0.5 [kg/cm²], it is adjusted by the flow rate predetermined with the mass flow rate controller 17, and is led to a reactor 11. 18 is the dielectric barrier electric discharge excimer lamp with which xenon gas was enclosed, and can generate the ultraviolet radiation UV with the main powerful wavelength 172 [nm]. Incidence of the ultraviolet exciting line by which outgoing radiation was carried out is carried out to the reactor 11 interior through the synthetic quartz aperture 19 from the excimer lamp 18. The exhaust air bulb which opens and closes the exhaust pipe arrangement with which 20 connects a vacuum pump 12 to a reactor 11, and 21 are atmospheric-exhaust bulbs. In addition, the wavelength of the above-mentioned ultraviolet radiation is the value of the range which has the capacity which decomposes oxygen and is made into active oxygen, and needs to be less than 190nm generally.

[0042] The following procedures perform UV/O₂ annealing. Opening of the exhaust air bulb 20 is carried out, and evacuation of the inside of a reactor 11 is immediately carried out at the same time it carries the composite-construction object 13 on the susceptor 14 currently heated by predetermined temperature and operates a vacuum pump 12. The exhaust air bulb 20 is embarrassed in the place where the internal pressure arrived at the base of 10^{-5} [Torr], a vacuum pump 12 is suspended, and high grade O₂ gas is introduced into a reactor 11 with the mass flow rate about 1000 [cc/min]. In the place where the internal pressure of a reactor turned into an ordinary pressure (= atmospheric pressure), opening of the atmospheric-exhaust bulb 21 is carried out, and O₂ surplus gas introduced henceforth is discharged out of a reactor 11 with a predetermined mass flow rate. A reactor 11 is maintainable in the pure oxygen atmosphere of atmospheric pressure by carrying out like this. Outgoing radiation of the ultraviolet

exciting line is continuously carried out to the reactor 11 interior from the excimer lamp 18. The ultraviolet radiation of the 172 [nm] wavelength which carried out incidence to the reactor 11 interior is strongly absorbed to O₂ in a reactor 11, and the active oxygen which consists of O₃, O (3P), O (1D), O₂ (1deltag), etc. generates it as a result. In the place where the above preparation was completed, UV/O₂ active-oxygen annealing is carried out predetermined time. Typical annealing conditions are shown below.

[0043]

----- Active oxygen annealing (UV/O₂) conditions Container pressure
 Atmospheric pressure Processing temperature 350 [**]
 Excitation intensity (synthetic quartz glass position) 13 [mW/cm²] (172 [nm])
 O₂ flow rate 200 [cc/min]
 Processing time 15 [min]

If ----- annealing is completed, the outgoing radiation of the excimer lamp 18 and introduction of a high grade O₂ will be stopped, and the composite-construction object 13 will be taken out immediately. In this way, UV/O₂ annealing finishes and the composite-construction object of the crystalline oxide dielectric thin film of this invention and a single crystal Si substrate is completed.

[0044] In addition, the above-mentioned processing temperature (annealing temperature) is made into the range from ordinary temperature to about about 500 degrees C, and its about 200-500 degrees C are especially desirable. According to the experimental result, active oxygen changes to usual oxygen at the elevated temperature of 500 degrees C or more, and annealing time becomes long below 200 degrees C.

[0045] Next, it explains, making the effect of the example of this invention contrast with the survey data of the conventional example. The characteristic curve (b) of drawing 7 shows the leakage-current (current which flows on YSZ film) property which impressed and measured positive voltage in the MIS capacity which carried out the vacuum evaporation of the aluminum gate electrode, and formed it on the YSZ film in the composite-construction object of the YSZ/Si substrate produced based on the above-mentioned example 1. In addition, a characteristic curve (a) is the leak property (previous statement) of the composite-construction object by the conventional example.

[0046] If (b) is compared with the characteristic curve (a) of drawing 7, it turns out that the example of this invention is reducing the leakage current by 4 figures - 5 figures in general compared with the conventional example. If the property of $E_{ox}=1$ [MV/cm] is seen, in the example of this invention, the value below $J=1$ [nA/cm²] is given to the conventional example being a current density $J=100$ [muA/cm²] base.

[0047] Next, although it was $BE_{ox}=1.5$ [MV/cm] in the conventional example according to measurement when its attention was paid to the insulating disruptive strength BE_{ox} , $BE_{ox}=4.6$ [MV/cm] and improvement in 3 or more times were found in this invention example.

[0048] moreover, the TDDb life when giving constant-current stress 0.1 [mA/cm²] -- the conventional example -- amount Q_{bd} of net charge = -- 1.2 [mC/cm²] -- it was (previous statement) -- receiving -- the example of this invention -- Q_{bd} = -- it is the value of 850 [mC/cm²] and the improvement of about 3 figures was obtained

[0049] From the above result, it is understood that this invention is very effective in reduction of (1) leakage current of the crystalline oxide thin film in the composite-construction object of a crystalline oxide thin film / Si substrate, improvement in (2) insulation disruptive strength, and impudence of (3) TDDb life.

[0050] (Example 2) The 2nd concrete example of the composite-construction object based on this invention is an example which forms the composite-construction object which consists of a single crystal Si substrate of a field (100), and a single crystal cerium-oxide (CeO₂) thin film 2 of the field (110) grown epitaxially using a high-vacuum electron-beam-evaporation method and UV/O₃ active-oxygen annealing. "/O₃" means the oxygen as a raw material containing 8% of ozone O₃ here.

[0051] First, n type the surface contamination and surface natural oxidation object of a single crystal Si substrate of a field (100) are removed by RCA washing and rare fluoric acid washing. PITAKISHARU growth of the CeO₂ thin film is continuously carried out by the high-vacuum electron-beam-evaporation

method on Si substrate. A vacuum evaporation machine uses the same thing as the equipment used by the aforementioned YSZ growth. A vacuum evaporation target is the tablet which fabricated and made CeO₂ powder of 99.999% or more of purity with the hotpress the form of a diameter 15 [mm] and thickness 7 [mm]. An evaporation source is filled up with CeO₂ tablet, Si substrate is installed in the substrate electrode holder of a vacuum evaporation tub, and the inside of a vacuum evaporation tub is exhausted. If the pressure of a vacuum evaporation tub amounts to 10^{-6} [Torr], substrate temperature will be heated and it will be made predetermined membrane formation temperature. Temperature is stabilized, a high-speed electron beam is made bombardment [in the place where tub internal pressure fell below in 10^{-9} [Torr] / a tablet], and CeO₂ film is grown up by the growth rate of 0.4 [nm/min]. Growth is continued until it gathers a growth rate to 1.55 [nm/min] and becomes predetermined thickness, while introducing high grade dryness oxygen into a vacuum evaporation tub in the place where thickness became 5 [nm] in general and adjusting a pressure to 8×10^{-6} [Torr]. A growth rate controls and adjusts the emission current of hot cathode. The typical membrane formation conditions from which single crystal CeO₂ thin film of the above-mentioned (110) field is obtained are as follows. [0052]

----- (110) CeO₂ membrane-formation conditions of a field Membrane formation pressure Below 1×10^{-8} [Torr] (up to thickness 3 [nm])

8×10^{-6} [Torr] (from thickness 3 [nm] to O₂ introduction)

Growth temperature 800 [°C] (substrate temperature)

Target 99.999% CeO₂ of purity tablet Substrate distance 30cm Electron beam acceleration voltage The output of a vacuum evaporation power supply is cut in the place which became the thickness of a 8kV -

----- request, and membrane formation is finished. Thus, CeO₂ deposited film was a single crystal epitaxial film of a field which has fluorite structure (cubic) (110). It is also possible to obtain single crystal CeO₂ film of the field where the directions of orientation incidentally differ on Si substrate of the same (100) field (111). In this case, growth temperature is set to 600 [°C] in the above-mentioned growth conditions. The conventional example is completed in this stage. With the composite-construction object based on this invention, an active oxygen annealing "UV/O₃ annealing" process is carried out after this.

[0053] Drawing 3 is the typical important section cross section of the equipment used for annealing. Since the function and composition of the part which attached the same number as drawing 2 are the same as drawing 2, redundancy is avoided and explanation is omitted. In drawing 3, 31 is CeO₂ / Si composite-construction object on the way of a process. The silent-discharge type ozonator 32 is installed between ready **** 16 and the mass flowmeter 17. This ozonator 32 can generate O₃ [eight mol %] for oxygen in a raw material. 33 is a low-pressure mercury lamp and can generate ultraviolet radiation with wavelength 185 [nm] and the bright line strong against 254 [nm]. Incidence of the ultraviolet exciting line by which outgoing radiation was carried out is carried out to the reactor 11 interior through the synthetic quartz aperture 19 from a low-pressure mercury lamp 33.

[0054] The procedure of UV/O₃ annealing is almost the same as the UV/O₂ above-mentioned annealing. That is, the structure 31 is carried on the susceptor 14 of predetermined temperature, and evacuation of the inside of a reactor 11 is carried out immediately. The exhaust air bulb 20 is embarrassed in the place where the pressure of the reactor 11 interior arrived at the base of 10^{-5} [Torr], a vacuum pump 12 is suspended, and high grade O₂ gas is introduced into a reactor 11 with the mass flow rate about 1000 [cc/min]. High grade O₂ gas which an ozonator 32 is operated and contains 8% of O₃ in the place where the internal pressure of a reactor 11 turned into an ordinary pressure (= atmospheric pressure) at the same time it carries out opening of the atmospheric-exhaust bulb 21 is introduced into a reactor 11. Between is not kept but outgoing radiation of the ultraviolet exciting line is carried out to the reactor 11 interior from a low-pressure mercury lamp 33. The ultraviolet radiation of 254 [nm] which carried out incidence to the reactor 11 interior photodissociates O₃ in a vessel, and the active oxygen which consists of O (1D), O₂ (1delta), O (3P), etc. besides O₃ generates it as a result. In this way, UV/O₃ active-oxygen annealing is carried out predetermined time. The typical annealing conditions are as follows.

[0055]

----- Active oxygen annealing (UV/O₃) conditions Container pressure

Atmospheric pressure Processing temperature 400 [**]

Excitation intensity (synthetic quartz glass position) 20 [mW/cm²] (254 [nm])O₂ flow rate 200 [cc/min]

Processing time 60 [min]

If ----- annealing is completed, the outgoing radiation of a low-pressure mercury lamp 33, the operation of an ozonator 32, and introduction of a high grade O₂ will be stopped, and a composite-construction object will be taken out immediately. In this way, the composite-construction object of the crystalline oxide dielectric thin film and single crystal Si substrate by this invention is completed.

[0056] Next, the effect of this example 2 is explained. The following table 1 is a table which compared the TDDB life (the amount Qbd of net charge) when giving leakage-current JL in the field strength 1 [MV/cm] of CeO₂ thin film (20 [nm]) of the CeO₂/Si composite-construction object based on the conventional example, the insulating disruptive strength BEox, and a constant current 0.1 [mA/cm²] to this example 2 row.

[0057]

[Table 1]

(表 1)

CeO₂/Si 複合構造体の電気特性の比較

	リーク電流 J L [n A / c m ²]	絶縁破壊強度 B Eox [M V / c m]	総電荷量 Q b d [m C / c m ²]
実施例	0.45	5.1	1800
従来例	3500	1.4	12.4

[0058] The electrical property of the above-mentioned table 1 impressed and measured positive voltage in MIS capacity (aluminum gate) like the aforementioned example 1. As [from Table 1] the Ming kana, the example 2 of this invention reduces a leakage current JL by 4 figures in general compared with the conventional example, and is raising the insulating disruptive strength BEox 3.5 times. Furthermore, the amount Qbd of net charge corresponding to a TDDB life was put on two 1 C/cm, and it succeeded in raising a TDDB life by 2 figures conventionally. That it is effective for the leakage current of the crystalline oxide thin film in the composite-construction object of a crystalline oxide thin film / Si substrate or an improvement of an insulating disruptive strength and a TDDB life has the clear example 2 of the above result to this invention.

[0059] In addition, in the above-mentioned active oxygen annealing, it is also possible to transpose to annealing which does not use UV, i.e., O₃ annealing. Since O₃ annealing also produces active oxygen, it is effective. Although the composition of a thermal treatment equipment becomes easy from UV/O₃, compared with the time of heat treatment time using UV, it is extended a double-precision grade. Moreover, other active oxygen annealing methods of the above-mentioned example or the below-mentioned example can be used.

[0060] (Example 3) The 3rd concrete example of the composite-construction object based on this invention is an example which forms the composite-construction object which consists of a single crystal Si substrate 1 of a field (100), and a single crystal strontium-titanate (SrTiO₃) thin film 2 of a

field (001) by the laser ablation vacuum deposition and plasma/O₂ annealing.

[0061] After removing n type the surface contamination and surface natural oxidation object of a single crystal Si substrate of a field (100) by RCA washing and rare fluoric acid washing, SrTiO₃ thin film is grown up in a high-vacuum laser ablation vacuum deposition on Si substrate. A vacuum evaporation machine is a general aviation equipped with two switchable sources of vacuum evaporation in the vacuum. Each source of vacuum evaporation is filled up with the SrO tablet and SrTiO₃ tablet which operated the powder of 99.999% or more of purity orthopedically in the shape of a disk (10phi, 10t) with the hotpress.

[0062] If the procedure of growth is explained, the Si substrate 1 immediately after the aforementioned washing will be installed in the substrate electrode holder in a vacuum evaporation tub, and the inside of a vacuum evaporation tub will be exhausted. This electrode holder has the capacity to heat a substrate to the temperature from ordinary temperature to 800 [°C], and has opposed on both sides of the distance of about 5 [cm] from the tablet. Exhaust air of vacuum evaporation equipment is powerful, and an internal pressure can lower back pressure to below 10⁻⁹ [Torr]. A substrate electrode holder is heated in the place where the pressure of a vacuum evaporation tub amounted to 10⁻⁸ [Torr], and it is made predetermined growth temperature. The preparations which use the 1st evaporation source with which SrO will be filled up by the time temperature is stabilized are made.

[0063] Turn on the output of a laser light source in the place where the pressure in a vacuum evaporation tub fell below in 10⁻⁸ [Torr], a high-density laser beam is made bombardment [a tablet], and growth of a very thin SrO buffer film is started. The laser of this example is a KrF (wavelength [of 248nm], 20ns of pulse width) excimer laser. Repeat frequency was 10 [Hz] and the energy density was 1.0 [J/cm²]. The thickness of a SrO buffer film suspends the output of laser by ***** set to 6 [nm], and stops growth. While changing an evaporation source to the 2nd evaporation source in which SrTiO₃ tablet is stored immediately, the temperature of a substrate electrode holder is changed into the predetermined temperature suitable for SrTiO₃ growth. In the place by which temperature was stabilized, high grade oxygen is introduced from the exterior, a pressure is set to 5x10⁻⁵ [Torr], the output of laser is turned on again, and SrTiO₃ film is grown up shortly. Dispatch of a laser light source is stopped in the place where thickness became a desired value, introduction of O₂ is stopped, and a substrate is annealed. It is as follows when an example of the growth conditions from which SrTiO₃ film of a single crystal (001) side is obtained is summarized.

[0064]

----- Growth conditions of a SrO buffer film Growth pressure Below 1x10⁻⁸ [Torr] Growth temperature 800 [°C] (substrate temperature)

Tablet-substrate distance 5 [cm]

Laser light source KrF excimer laser (energy 130mJ)

Excitation pulse 20ns of width of face, repeat frequency 10Hz----- Growth conditions of SrTiO₃ film Growth pressure 5x10⁻⁵ [Torr] (99.999% oxygen introduction)

Growth temperature 600 [°C] (substrate temperature)

Other conditions take out a substrate from a vacuum evaporation tub, after the same ----- substrate temperature as upper SrO becomes sufficiently low. the SrO layer of 6 [nm] formed first disappears in process in which SrTiO₃ next film is grown up (it is thought that it deteriorated on SrTiO₃ film) -- it is -- the film formed in this way is single crystal SrTiO₃ monolayer as a matter of fact

[0065] The conventional example is completion with the composition of this stage. With the composite-construction object based on this invention, active oxygen annealing is performed after this. Although the UV/O₂ above-mentioned annealing and UV/O₃ annealing are also possible, the example of "plasma/O₂ annealing" is introduced here.

[0066] Drawing 4 is the typical important section cross section of the equipment used for plasma/O₂ annealing. In drawing 4 , the reactor with which, as for 41, water cooling of the side attachment wall is carried out, and 42 are the vacuum pumps for exhausting a reactor 41, and can maintain the pressure of a reactor 41 from an ordinary pressure before 10⁻⁶ [Torr]. 43 is a composite-construction object in front

of annealing. 44 is the susceptor equipped with a heating means to support the composite-construction object 43 and to hold to predetermined temperature. This susceptor 44 is grounded electrically. The composite-construction object 43 is put on a susceptor 44 so that SrTiO₃ film surface may be upwards suitable. The power electrode 45 is placed so that a susceptor 44 may be opposed. The power electrode 45 is electrically connected to RF generator 48 (for example, frequency of 13.56MHz) besides a vessel through the blocking capacity 46 and the adjustment machine 47. 49 is the chemical cylinder filled up with oxygen O₂ of 99.999% or more of purity. After 49 chemical cylinder O₂ is decompressed by ready **** 50 by the pressure about 0.5 [kg/cm²], it is adjusted by the flow rate predetermined with the mass flow rate controller 51, and is led to a reactor 41. The exhaust air bulb which opens and closes the exhaust pipe arrangement with which 52 connects a vacuum pump 42 to a reactor 41, and 53 are pressure-control bulbs which control the pressure of a reactor uniformly.

[0067] The procedure of plasma/O₂ annealing is as follows. Opening of the exhaust air bulb 52 is carried out, and evacuation of the inside of a reactor 41 is immediately carried out at the same time it carries the composite-construction object 43 before processing on the susceptor 44 currently heated by predetermined temperature and operates a vacuum pump 42. While introducing high grade O₂ gas into a reactor 41 with a predetermined mass flow rate in the place where the pressure of the reactor 41 interior arrived at the base of 10~5 [Torr], the pressure-control bulb 53 is operated, and the internal pressure of a reactor 41 is adjusted to a predetermined value. After waiting until internal pressure and susceptor temperature are stabilized, turn on the output of RF generator 48, electric discharge is made to cause between a susceptor 44 and the power electrode 45, and annealing is started. If electric discharge takes place, the active oxygen which consists of O₃, O (3P), O (1D), O₂ (1deltag), etc. will occur. In the place where the above preparation was completed, UV/O₂ active-oxygen annealing is carried out predetermined time. Typical annealing conditions are shown below.

[0068]

----- Active oxygen annealing (plasma/O₂) conditions Container pressure 3

[mTorr]

Processing temperature 400 [**]

Susceptor-power inter-electrode distance 30 [cm]

O₂ flow rate 1 [cc/min]

RF power 100 [W]

Processing time 20 [min]

Annealing will be ended if ----- predetermined time passes. That is, a RF output is suspended, the operation of the pressure-control bulb 53 is stopped, the exhaust air bulb 52 is closed, a vacuum pump 42 is suspended, air is introduced into a reactor 41, air opening is carried out, and the composite-construction object 43 is taken out of a vessel. Such is carried out and the composite-construction object of single crystal SrTiO₃ film and a single crystal Si substrate is completed.

[0069] Next, the effect of the example 3 of this invention is explained. The following table 2 is a table which measured the amount Q_{bd} of net charge when giving leakage-current J_L in the field strength 1 [MV/cm] of SrTiO₃ thin film (40nm), the insulating disruptive strength BE_{ox}, and a constant current 0.1 [mA/cm²] with the example 3 row in the SrTiO₃/Si composite-construction object based on the conventional example.

[0070]

[Table 2]

(表 2)

S r T i O₃ / S i 複合構造体の電気特性の比較

	リーク電流 J L [n A / c m ²]	絶縁破壊強度 B E o x [M V / c m]	総電荷量 Q b d [m C / c m ²]
実施例	0.089	4.9	1350
従来例	220	2.3	8.1

[0071] The electrical property of Table 2 impressed and measured positive voltage in MIS capacity (aluminum gate) like the aforementioned example. As shown in Table 2, in an example 3, compared with the conventional example, a leakage current J_L is reduced 3-figure strength, and the insulating disruptive strength B E o x is raised to double precision. Furthermore, the amount Q_{b d} of net charge corresponding to a TDDb life was increased by 3 figures, and it succeeded in putting Q_{b d} on the base of 1 [C/cm²]. It is shown that the above result is the technology which the example 3 of this invention can greatly contribute to the leakage current of a crystalline oxide thin film and the improvement of reliability in the composite-construction object of a crystalline oxide thin film / Si substrate.

[0072] the effect of improving the leakage current and reliability of a crystalline oxide thin film of such a composite-construction object is an effect of appearing equally also to other single crystal oxide thin films and stacking-tendency oxide thin films which are not limited to the YSZ film explained in the concrete example to here, CeO₂ film, and SrTiO₃ film, and were grown up into Si single crystal substrate, and the scope of this invention is wide However, the optimum conditions of active oxygen annealing differ according to the target composite-construction object.

[0073] Moreover, on account of explanation, although all were made into the monolayer, as for the composition of a crystalline oxide thin film, an improvement effect with the same said of the cascade screen of a film of a different kind is acquired. For example, the remarkable leak mitigation effect, an insulating disruptive strength, and a TDDb resistance improvement effect show up also with the CeO₂/SrTiO₃/Si composite-construction object which carried out the laminating of the CeO₂ single-crystal film of a field (001), and the SrTiO₃ single-crystal film of a field (001).

[0074] Next, the example which applied this invention to the electronic device is explained.

(Example 4) this example is an example which applied information-separator composite-construction object shown in aforementioned drawing 1 to the MFIS type nonvolatile memory cell.

[0075] Drawing 5 is the typical structure section view of a MFIS type nonvolatile random access memory cell. Although this cellular structure is the same as the conventional cellular structure apparently explained by aforementioned drawing 8, the places whose crystalline oxide dielectric thin films I are films to which active oxygen annealing was given differ greatly. In drawing 5, the place in which 61 is the single crystal Si substrate of a p type field (100) and the thick field oxide film by which 62 was alternatively formed on Si substrate, and this field oxide film 62 is not formed is the MORT field 63. The memory cell is formed in this MORT field 63. 64 is the single crystal oxide dielectric film by which hetero-epitaxial growth was carried out to the MORT field (I), for example, (100), the YSZ film of a field, and is the film to which active oxygen annealing was given by the time the element was completed. On the single crystal oxide dielectric film 64, the ferroelectric film 65, for example, (001), PbTiO₃ film of a field, and the gate electrode 66, for example, Pt electrode, are. The ferroelectric film 65 is carrying out hetero-epitaxial growth on the crystalline oxide dielectric thin film 64. 67 and 68 are n type impurities, for example, the source electrode and drain electrode by which P and As are doped by

high concentration.

[0076] Next, if the main point of the manufacturing process of this element is explained briefly, after removing the contamination of the front face of the single crystal Si substrate of a p type field (100) by RCA washing first, the thermal oxidation film of 40 [nm] will be formed in the front face of Si substrate by the oxidizing [thermally] method, and Si₃N₄ film (it is written as a SiN film below) of 150 [nm] will be further formed by the LPCVD method on it. It continues, leaves the portion equivalent to the MORT field using phot lithography and the plasma etching method, and a SiN film (and thermal oxidation film) is removed. Next, after exfoliating and carrying out RCA washing of the resist again, the field oxide film of 400 [nm] is formed by the oxidizing [thermally] method (LOCOS oxidization). At this time, since an oxide film does not grow up to be the portion in which the SiN film is formed, the MORT field is also formed in it together with a field oxide film. Then, heat phosphoric acid removes a SiN film.

[0077] It continues and the YSZ film of a single crystal (001) side is grown up to be the MORT field. After removing the contamination and natural oxidation film of the MORT field by RCA washing and cleaning with diluted hydrofluoric acid, the YSZ film which contains Y₂O₃ [about 13 mol %] all over a substrate is grown up in a 20[nm] high truth atmospherics beam vacuum deposition. Since the example of the above-mentioned composite-construction object explained this membrane formation, explanation is not repeated here. After growth of a YSZ film finishes, a substrate is made immersed in the rare fluoric acid of concentration for about 10 seconds 1%, and by ultrapure water, a rinse is carried out and it dries.

[0078] In the place which washing by rare fluoric acid finished, the single crystal PbTiO₃ of a field (001) is grown up using the MOCVD (organic-metal vapor growth) method. The raw materials to be used are the tetraethyl lead [Pb (C₂H₅)₄], and tetraisopropoxy titanium [Ti (i-OC₃H₇)₄] and dryness oxygen (O₂). Front 2 persons are liquids in an ordinary temperature ordinary pressure, evaporate all with a carburetor and convey them to a reactor by dryness Ar carrier gas. Growth equipment uses not special equipment but a general aviation. The typical membrane formation conditions are as follows.

[0079]

----- (001) MOCVD growth conditions of the single crystal PbTiO₃ of a field

Growth pressure 6 [Torr]

Growth temperature 640 [**] (substrate temperature)

Field Charge Pb (C₂H₅)₄, Ti (i-OC₃H₇)₄, O₂ Evaporation temperature -15 degrees C [Pb (C₂H₅)₄]

25 degrees C [Ti (i-OC₃H₇)₄]

A quantity of gas flow O₂ 100 cc/min Pb raw material carrier (Ar) 25 cc/min Ti raw material carrier (Ar) 22 cc/min Ar dilution 200cc/min----- a degree -- active oxygen annealing, for example, UV/O₃ annealing, -- predetermined time operation -- it carries out detailed explanation of UV/O₃ annealing is already described -- it comes out and omits The leakage-current property of not only a YSZ film but the three or so PbTiO₃(s) dielectric film which grew previously, and an insulating disruptive strength and TDD resistance are simultaneously improvable with this active oxygen annealing. In addition, even if it performs active oxygen annealing immediately after YSZ film formation, although it is good of course, an improvement of the electrical property of a ferroelectric film cannot be desired in this case.

[0080] If growth of a ferroelectric film is ended, next, complete membrane formation of the gate electrode material Pt, for example, platinum, will be performed. Membrane formation of Pt uses the general-purpose DC magnetron sputtering method. As an electrode material, Pt target more than 99.99% purity and thickness are 100 [nm] **. The typical sputtering membrane formation conditions of Pt are as follows.

[0081]

----- DC MAKUNE TRON sputtering conditions of Pt Membrane formation

pressure 3 [mTorr]

Membrane formation temperature 400 [**] (substrate temperature)

Field Charge 99.99%Pt target Spatter gas Ar Target-substrate distance 30 [cm]

DC power 150 [W]

----- if membrane formation of Pt ends, Pt gate electrode will be formed using a photolithography and Ar sputter-etching technology. The resist which etching of Pt ended is left as it is, and reactive ion etching of PbTiO₃ film and a YSZ film is performed continuously. Etching gas is the mixed gas of CF₄ and O₂. If etching of a YSZ film is completed, a resist will be made to ash, it will remove and the dust on the front face of a substrate will be removed by ultrasonic cleaning using the organic solvent.

[0082] It continues and the ion implantation of the P is carried out all over a substrate. The portion except the gate electrode of the MORT field, i.e., the source, and a drain field are poured into Si substrate. Acceleration voltage is 100 [keV] and a dose is 5×10^{15} [an individual / cm²]. If an ion implantation is completed, activation of the source and the pouring impurity of a drain will be performed using rapid heat treatment (RTA) equipment. Heat treatment temperature is 1200 [°C] and heat treatment time is 30 seconds. Thus, a MFIS type ferroelectric memory cell is completed.

[0083] (Example 5) Other examples of an electronic device are examples in which the MOS transistor was formed on the SOI substrate which used information-separator composite-construction object of the aforementioned this invention. Drawing 6 is the typical structure section view of the MOS transistor of SOI structure. The places whose crystalline oxide dielectric thin films pinched between Si epitaxial film and Si single crystal substrate are films to which active oxygen annealing was given although this structure is the same as the structure of the conventional example where it explained by drawing 9, seemingly differ greatly.

[0084] In drawing 6, it is the film to which active oxygen annealing was given by the time the single crystal Si substrate of the field (100) of n form where the impurity dope of 71 was carried out at high concentration, and 72 are the single crystal oxide dielectric films by which hetero-epitaxial growth was carried out on the Si substrate 71 (I), for example, (100), the YSZ film of a field, and the element completed them. 73 is n form which carried out hetero-epitaxial growth on the YSZ film 72, or the single crystal Si film of the comparatively thick (100) field of p form. The thick field oxide film by which 74 was formed on the single crystal Si film 73, and 75 are the MORT fields. 76 in the MORT field 75 is the poly silicon-gate electrode, and the SiO₂ thermal-oxidation film (gate oxide film) 77 is formed between this and the single crystal Si film 73. Moreover, 78 and 79 are the source electrodes and drain electrodes by which p form or the impurity of n form is doped by high concentration.

[0085] If the main point of a manufacturing process is explained briefly, a front face will be first made pure for the single crystal Si substrate 61 of the field (100) of n form doped by high concentration by RCA washing and rare fluoric acid washing, and the YSZ film of a field (001) will be grown up all over the in 150 [nm] and the already explained high truth atmospherics beam vacuum deposition. Active oxygen annealing explained to this information-separator composite-construction object above, for example, plasma/O₂ annealing, is given a little longer after growth. Then, a substrate is made immersed in the rare fluoric acid of concentration for about 10 seconds 1%, and by ultrapure water, a rinse is carried out and it dries. The front face of the YSZ film 72 of a field (100) is continuously grown epitaxially by the ordinary-pressure CVD of common knowledge of the single crystal Si of a field (100) with a thickness of 2 micrometers. Raw materials are SiCl₄ and H₂. Thus, the SOI substrate of a low leakage current and high dielectric-breakdown resistance is made.

[0086] Next, the 3×10^{13} [individual / cm²] ion implantation of the impurity B of p form is carried out all over an epitaxial Si film, and a substrate front face is made pure by RCA washing. The field oxide film 74 and the MORT field 75 are formed by the completely same LOCOS oxidation style as the above-mentioned example. While passing through a LOCOS oxidization process, in an epitaxial Si film, B impurity which carried out the ion implantation previously is diffused widely, and is activated.

[0087] After continuing, growing up the thin SiO₂ thermal-oxidation film 77 (thickness 20 [nm]) which oxidizes a substrate thermally and turns into a gate oxide film to the MORT field and depositing the polysilicon contest film of thickness 300 [nm] by the LPCVD method on this further, as shown in drawing 6, the poly silicon-gate electrode 76 is formed using a photolithography and reactive ion etching.

[0088] After removing a resist, the ion implantation of the P is carried out to a substrate front face. It is the portion except the gate electrode of the MORT field, i.e., the source, and a drain field that B is driven into an epitaxial Si film by this pouring. B is simultaneously driven also into the poly silicon-gate electrode. Acceleration voltage is 100 [keV] and a dose is 5×10^{15} [an individual / cm²].

[0089] In order to activate the impurity of the source electrode 78, the drain electrode 79, and the gate electrode 76 after making a substrate front face pure by RCA washing if an ion implantation is completed, the drive-in of 2 hours is performed within the diffusion furnace heated to 1120 [°C]. Thus, the MOS transistor using the SOI substrate based on this invention is completed.

[Translation done.]

* NOTICES *

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1. This document has been translated by computer. So the translation may not reflect the original precisely.
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 PRIOR ART

[Description of the Prior Art] Generally the single crystal film has the feature that many physical properties flat in crystal structure and peculiar to the material, such as eye a uniform hatchet, conductivity, a dielectric, insulation, a ferroelectricity, superconductivity nature, a semiconductor property, light-transmission nature, optical magnetic properties, and crystal periodicity, appear keenly strongly, compared with the amorphous film or the polycrystal film. Therefore, it is useful in various industrial fields, and the realization is desired strongly. In recent years, these single crystal film is loaded into Si semiconductor device or an integrated circuit, and the attempt which is going to realize the high integrated circuit and high smart device of the added value which cannot be attained only with the conventional Si system material is performed actively, and is come and required. Basic structure indispensable when such a new Si semiconductor device is constituted is the composite-construction object (the following information-separator structure and abbreviated name) of a crystalline dielectric thin film (I) and a single crystal Si (S).

[0003] When an example is given, there is a MFIS type nonvolatile random access memory cell as shown in the typical structure section view of drawing 8 . This cell replaces O portion of an MOS transistor which consists of a (Metal M)-silicon-oxide (O)-semiconductor (S) by the single crystal ferroelectric film (F) which has electrically the spontaneous polarization which can be reversed, and the single crystal dielectric film (I).

[0004] In drawing 8 , it is the single crystal oxide dielectric film (I) by which 101 was carried out at the single crystal Si substrate of a p type field (100), and hetero-epitaxial growth of 102 was carried out on the Si substrate 101, for example, (100) is the YSZ film of a field. 103 is the ferroelectric film by which hetero-epitaxial growth was carried out on the single crystal oxide dielectric film 102, for example, (001) is PbTiO₃ film of a field. 104 is the gate electrode formed on the ferroelectric film 103, for example, is Pt electrode. 105 and 106 are n type source electrodes and drain electrodes, they carry out the ion implantation of P (Lynn) or the As (arsenic), respectively, make it activated by thermal diffusion, and are formed.

[0005] By this memory, information is recorded by impressing a positive or negative pulse voltage between gate (electrode G)-Si substrates (Sub), carrying out reversal fixation of the spontaneous-polarization vector of a ferroelectric film, and making a transistor into a flow or non-switch-on.

[0006] Moreover, as the above-mentioned ferroelectric layer, PbTiO₃ (lead titanate) film formed by the chemical vapor growth (CVD), the magnetron sputtering method, etc., Pb(Zrx, Ti 1-x) O₃ (zircon lead titanate) film, 3OBi₄Ti₁₂ (titanic-acid bismuth) film, etc. are examined (all are polarization shaft orientation single crystal films).

[0007] Moreover, as the above-mentioned single crystal dielectric layer, CeO₂ (cerium oxide) film formed by the electron-beam-evaporation method or CVD, a YSZ (ITTORIYA stabilization JIRUKONIYA) film, SrTiO₃ (strontium titanate) film, etc. are examined. These single crystal dielectric films are bearing a role of a buffer coat which prevents that function as a template layer for forming a single crystal ferroelectric film, and also Si substrate carries out counter diffusion to a ferroelectric.

[0008] Moreover, MFMIS structure is also proposed by the nonvolatile random access memory cell

besides the above-mentioned MFIS structure. Single crystal F layer and M layers inserted between I layers of single crystals are here, the electric conduction films, for example, Pt film etc., of a single crystal etc. information-separator structure is used for the lower part of MFM structure also in this case. [0009] Next, the MOS transistor of the SOI structure shown in the typical structure section view of drawing 9 is also the important application of information-separator structure. The MOS transistor of this structure has the feature which should be mentioned especially that it can escape from a very detrimental latch up completely on operation.

[0010] In drawing 9, they are single crystal dielectric films, such as CeO₂ by which hetero-epitaxial growth was carried out by 111 touching Si substrate of a field (100) and 112 touching this Si substrate 111, YSZ, and SrTiO₃. This Si substrate 111 and the single crystal dielectric film 112 have constituted information-separator structure. The single crystal Si layer of the p type field (100) where hetero-epitaxial growth of 113 was carried out by CVD on the single crystal dielectric film 112, the SiO₂ gate oxide film in which 114 was formed by thermal oxidation of the single crystal Si layer 113, and 115 are the gate electrodes of contest polysilicon formed by CVD and dry etching on the gate oxide film 114. 116 and 117 are the source electrodes and drain electrodes of N type, they carry out the ion implantation of P (Lynn) or the As (arsenic), respectively, make it activated by thermal diffusion, and are formed. [0011] Moreover, the oxide high-temperature superconductivity thin film wiring tried aiming at high speed processing of an integrated circuit also forms quality single crystal superconductor film (for example, YBa₂Cu₃O₇) wiring on information-separator structure. As I layers, YSZ and the single crystal cascade screen of Y₂O₃ are used, for example.

[0012] Thus, it is understood that information-separator structure is the important basic structure for realizing various devices containing the functional thin film (a ferroelectric film, a semiconductor film, an electric conduction film, a superconductivity film, dielectric film) of a single crystal on a single crystal Si substrate. However, in the present condition, the problem that a low and (3) passage-of-time destructive (TDDB) life have short (2) disruptive strengths with large (1) leakage current is in I layers of single crystal oxide dielectric thin films, and it has been a serious obstacle at realization of the above highly efficient devices.

[0013] The above-mentioned problem is explained in detail based on actual data. Drawing 7 is the leakage-current property view of a crystalline oxide dielectric thin film. It is that a characteristic curve (a) indicates the example of a property of the conventional single crystal YSZ film to be in drawing 7. On the YSZ film of the field (100) which grew epitaxially Si substrate of an N type single crystal (100) side Leakage-current density J [A/cm²] which impressed and measured positive voltage to the gate electrode of the MIS capacity which formed and produced aluminum electrode (gate) of a diameter 200 [μm] is shown as a function of field strength E_{ox} [V/cm]. In addition, in order to remove an absorption current component, measurement of current is performed, after carrying out voltage impression and time passes enough. Moreover, thickness is 40 [nm] and the forming-membranes method is an electron-beam-evaporation method here. The forming-membranes methods (vacuum evaporation conditions etc.) are later mentioned in the example of this invention in the cleaning-method row of a substrate.

[0014] The leakage current of the single crystal YSZ film of the conventional example is in a very high level so that the characteristic curve (a) of drawing 7 may be seen and may be known. Considering practical use, I want to suppress leakage-current density below to $J = 1$ [nA/cm²] at the time of field strength $E_{ox} = 1$ [MV/cm]. However, in the above-mentioned conventional example, this and the value which was greatly different widely are indicated to be the bases of $J = 100$ [μA/cm²]. Moreover, the current jump of A points in the characteristic curve (a) of drawing 7 shows that the YSZ film is carrying out dielectric breakdown on this voltage ($E_{ox} \approx 1.5$ [MV/cm]). In order to apply to the above-mentioned MFIS nonvolatile memory cell etc., it is the value which cannot be said that this insulating disruptive strength is enough.

[0015] It is indispensable conditions that there is long-term dielectric-breakdown (TDDB) resistance over power supply stress in I more layers. This resistance measures and evaluates the amount Q_{bd} of net charge [C/cm²] which passed the film by the time it generally gave and carried out dielectric breakdown of the constant-current stress to the dielectric film. The typical TDDB life when giving the constant-

current stress of $J = 0.1$ [mA/cm²] to the above-mentioned single crystal YSZ film was $Q_{bd} = 1.2$ [mC/cm²], respectively. This value is low comparable, if it compares with typical value $Q_{bd} = 10$ [C/cm²] of thermal oxidation SiO₂ film used as gate oxide films, such as an MOS transistor. In order for power supply stress to apply a YSZ film to a comparatively strong MFIS nonvolatile memory cell etc., the TDDB life of the base of at least 0.1 [C/cm²] is required.

[0016] Such a problem of I layers in information-separator structure is not having restricted to the single crystal YSZ film formed by the electron-beam-evaporation method, but is a problem by which the YSZ film produced by other single crystal oxide dielectric-materials film and other forming-membranes methods is also observed in common.

[Translation done.]

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DESCRIPTION OF DRAWINGS

[Brief Description of the Drawings]

[Drawing 1] The cross section showing the basic structure of the composite-construction object based on this invention.

_____ The cross section showing an example of the active oxygen annealer used by the manufacture method of this invention.

[_____] The cross section showing other examples of the active oxygen annealer used by the manufacture method of this invention.

_____ The cross section showing other examples of the active oxygen annealer used by the

[_____] The cross section showing the structure of a nonvolatile memory cell as an example of the electronic device of this invention.

_____ The cross section showing the structure which formed the MOS transistor on the SOI

| _____ | | | | _____ | |

[_____] The property view showing the leakage-current property of a crystalline oxide dielectric thin film.

_____ The cross section of an example of the nonvolatile memory cell based on the conventional electronic device structure.

[_____] _____ |
device structure.

[Description of Notations]

- 1 -- Single crystal Si substrate 2 -- Crystalline oxide dielectric film
- 11 -- Reactor 12 -- Vacuum pump
- 13 -- YSZ/Si composite-construction object in front of annealing
- 14 -- Susceptor 15 -- Oxygen cylinder
- 16 -- Ready **** 17 -- Mass flowmeter
- 18 -- Dielectric barrier electric discharge excimer lamp
- 19 -- Synthetic quartz aperture 20 -- Exhaust air bulb
- 21 -- Atmospheric-exhaust bulb 22 -- Evacuation equipment
- 31 -- Before [annealing] CeO₂/Si composite-construction object
- 32 -- Ozonator 33 -- Low-pressure mercury lamp
- 41 -- Reactor 42 -- Vacuum pump
- 43 -- Before [annealing] SrTiO₃/Si composite-construction object
- 44 -- Susceptor 45 -- Power electrode
- 46 -- Blocking capacity 47 -- Adjustment machine
- 48 -- RF generator 49 -- Oxygen cylinder
- 50 -- Ready **** 51 -- Mass flow rate controller
- 52 -- Exhaust air bulb 53 -- Pressure-control bulb
- 61 -- Si substrate 62 -- Field oxide film



Drawing 2
Drawing 3
Drawing 4
Drawing 5
Drawing 6

63 -- M O R T f i e l d 64 -- S i n g l e c r y s t a l d i e l e c t r i c t h i n f
65 -- F e r r o e l e c t r i c f i l m 66 -- G a t e e l e c t r o d e
67 -- S o u r c e e l e c t r o d e 68 -- D r a i n e l e c t r o d e
71 -- S i s u b s t r a t e 72 -- S i n g l e c r y s t a l d i e l e c t r i c t h i n f
73 -- S i n g l e c r y s t a l S i f i l m 74 -- F i e l d o x i d e f i l m
75 -- M O R T f i e l d 76 -- T h e p o l y s i l i c o n - g a t e e l e c t r o d e
77 -- G a t e o x i d e f i l m 78 -- S o u r c e e l e c t r o d e
79 -- D r a i n e l e c t r o d e

////////////////////////////////////

[T r a n s l a t i o n d o n e .]

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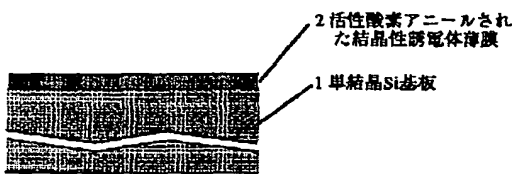
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DRAWINGS

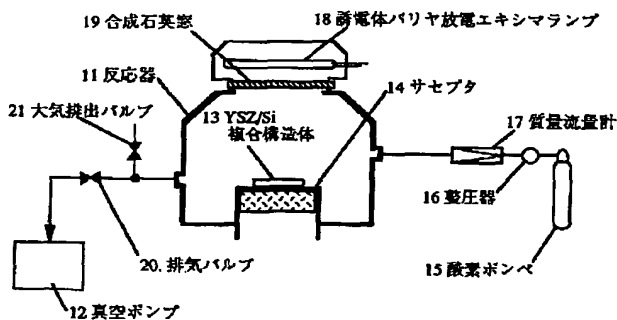
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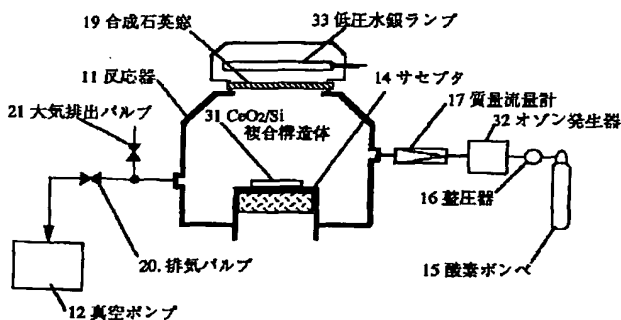
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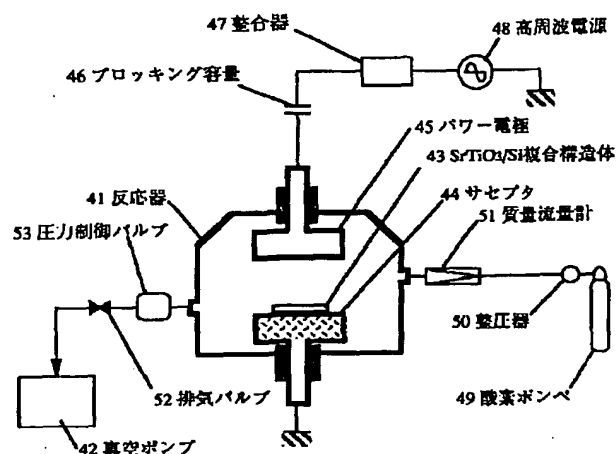
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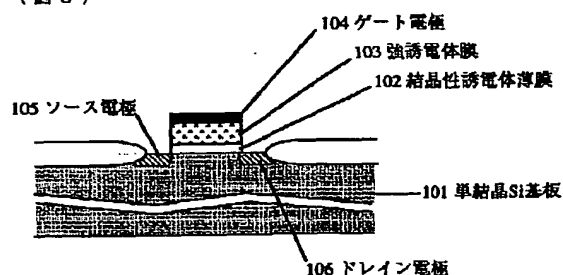
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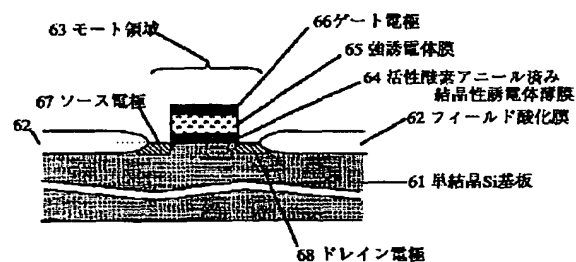
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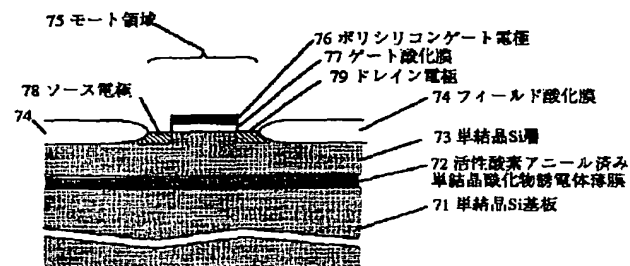
[Drawing 5]

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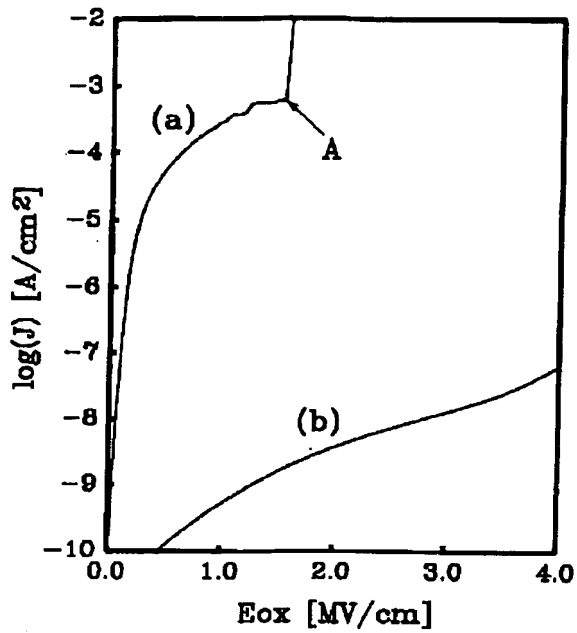


[Drawing 6]

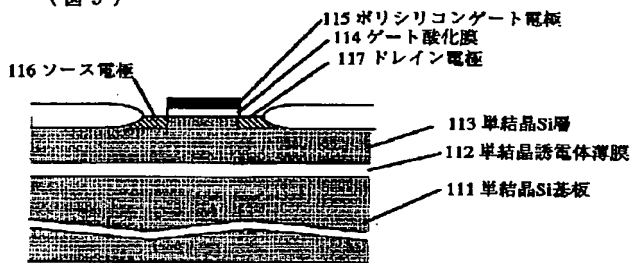
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[Drawing 7]
(図 7)



[Drawing 9]
(図 9)



[Translation done.]

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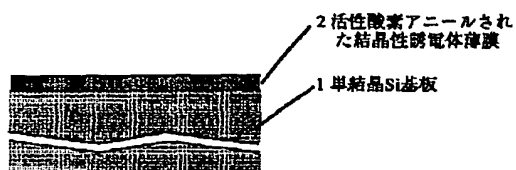
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DRAWINGS

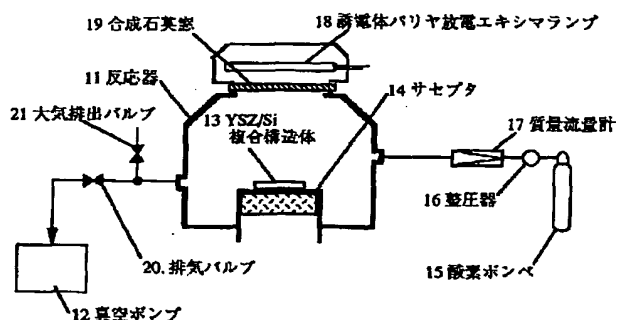
[Drawing 1]

(図 1)



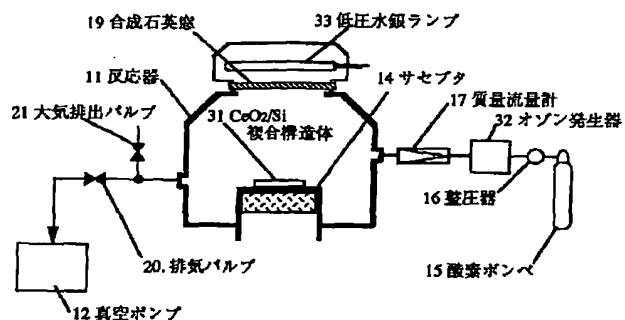
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(図 2)



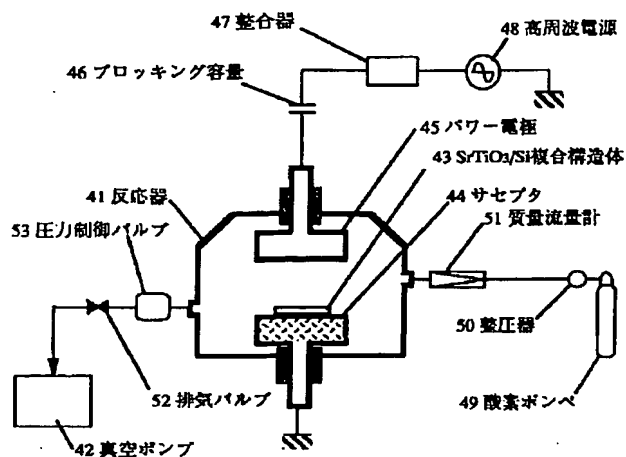
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(図 3)



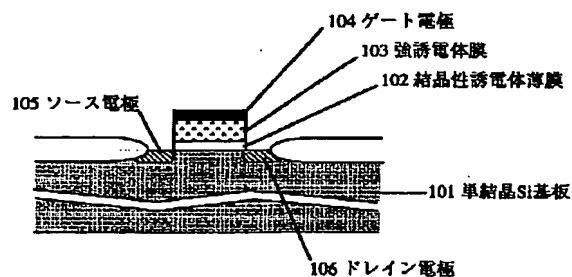
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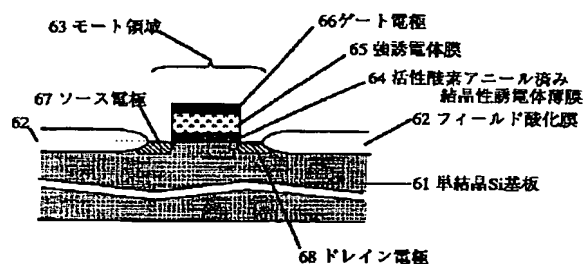
[Drawing 8]

(図 8)



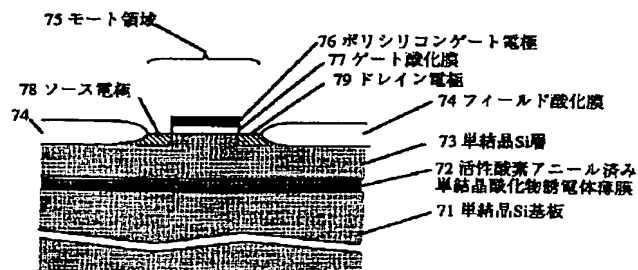
[Drawing 5]

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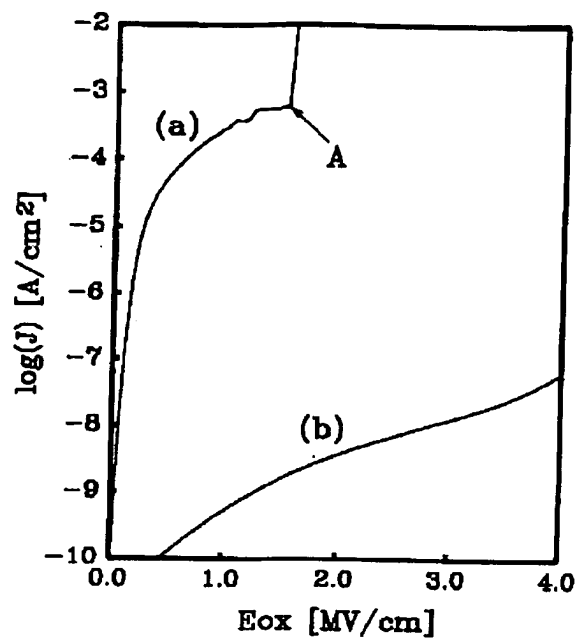


[Drawing 6]

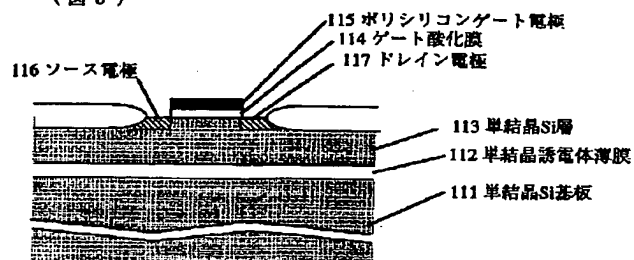
(図 6)



[Drawing 7]
(図 7)



[Drawing 9]
(図 9)



[Translation done.]